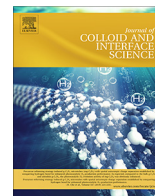




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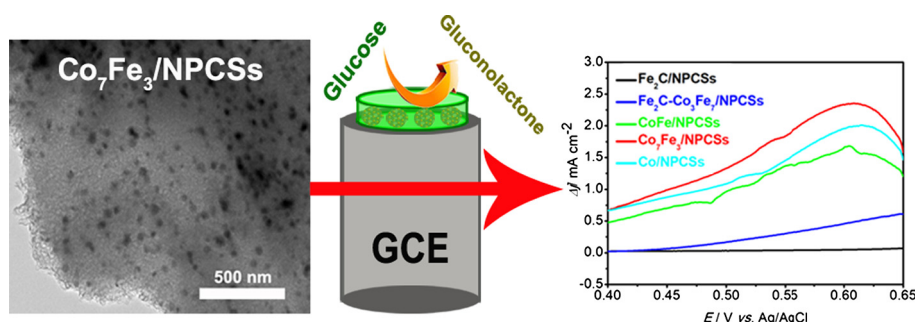
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Regular Article

Facile design of ultrafine Co_7Fe_3 nanoparticles coupled with nitrogen-doped porous carbon nanosheets for non-enzymatic glucose detectionMian Li^{a,*}, Jirong Yang^a, Mingjiao Lu^a, Yingjie Zhang^{a,*}, Xiangjie Bo^{b,*}^a National and Local Joint Engineering Laboratory for Lithium-ion Batteries and Materials Preparation Technology, Key Laboratory of Advanced Battery Materials of Yunnan Province, Faculty of Metallurgical and Energy Engineering, Kunming University of Science and Technology, Kunming 650093, China^b Key Laboratory of Nanobiosensing and Nanobioanalysis at Universities of Jilin Province, Department of Chemistry, Northeast Normal University, Changchun, Jilin Province 130024, China

GRAPHICAL ABSTRACT



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ABSTRACT

Ultrafine Co_7Fe_3 nanoparticles embedded in nitrogen-doped porous carbon nanosheets (denoted as $\text{Co}_7\text{Fe}_3/\text{NPCSS}$) are successfully synthesized by utilizing porous plant tissue as a precursor. The morphological, structural, and chemical contents of the $\text{Co}_7\text{Fe}_3/\text{NPCSS}$ and other control samples are analyzed by X-ray (powder) diffraction (XRD), scanning electron microscopy (SEM), transmission electron microscopy (TEM), N_2 adsorption–desorption, Raman spectroscopy, inductively coupled plasma (ICP), and X-ray photoelectron spectroscopy (XPS) techniques. The glucose oxidation and detection performances of each catalyst are evaluated by using cyclic voltammetry and chronoamperometry. The experimental results demonstrate that the electrocatalytic abilities of the resultant catalysts toward glucose oxidation decrease in the order of $\text{Co}_7\text{Fe}_3/\text{NPCSS}$ > Co/NPCSS > CoFe/NPCSS > $\text{Fe}_2\text{C}-\text{Co}_3\text{Fe}_7/\text{NPCSS}$ > $\text{Fe}_2\text{C}/\text{NPCSS}$. The experimental results prove that a small number of Fe atoms in Co_7Fe_3 can increase the number of active Co^{4+} sites. Meanwhile, the ultrafine Co_7Fe_3 nanoparticles uniformly dispersed along the porous carbon nanosheets' surfaces, which further improved the dispersion of the abundant electrochemically available active sites. Due to the synergistic effect of the hierarchical porous structures, high-density active sites and excellent electron conductivity, the optimal $\text{Co}_7\text{Fe}_3/\text{NPCSS}$ display the best glucose detection efficiency of all the catalysts examined. For instance, the $\text{Co}_7\text{Fe}_3/\text{NPCSS}$ exhibit large sensitivity values ($795.28 \mu\text{A cm}^{-2} \text{ mM}^{-1}$ between 0.001 and 2.20 mM and $401.98 \mu\text{A cm}^{-2} \text{ mM}^{-1}$ between 2.20 and 14.00 mM), a rapid response time (2.2 s), a low detection limit ($1.0 \mu\text{M}$), excellent anti-interference toward electroactive molecules, a perfect reproducibility and a superior long-term stability. The $\text{Co}_7\text{Fe}_3/\text{NPCSS}$ also exhibit a satisfying efficiency for glucose detection in human serum samples.

* Corresponding authors.

E-mail addresses: lim148@kmust.edu.cn (M. Li), zyjkmust@126.com (Y. Zhang), baoxj133@nenu.edu.cn (X. Bo).

Finally, our low-cost synthetic strategy can advance research used for designing 3D hierarchical meso/macroporous noble-metal-free catalysts without any tedious steps or templates.

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1. Introduction

It is well known that more than 300 million patients are suffering from diabetes, which therefore has been identified as one of the most serious health concerns of the 21st century [1]. Medical research has revealed the cause of diabetes, in which an insulin deficiency ascribed to a metabolic disorder of the body results in obvious hyperglycemia. In general, diabetes can be diagnosed by detecting patients' blood glucose concentrations and comparing the test results with the normal glucose concentration ranges of a healthy person ($80\text{--}120\text{ mg dL}^{-1}$) [2]. Therefore, developing advanced glucose sensors with high detection performances is extremely pivotal for diagnosing and managing diabetes mellitus. As glucose oxidase (GO_x) can efficiently catalyze the glucose oxidation reaction, Karl Erik Hellström and Inggerd Hellström have successfully designed the first GO_x -based sensor for glucose concentration detection [3]. Since then, GO_x -based glucose sensors have been extensively studied and improved to exhibit high sensitivities and selectivities. Nevertheless, enzymes are often unstable, and thus, GO_x -based glucose sensors inevitably suffer from worsening long-term stabilities [4].

Based on the existing facts referring to GO_x -based glucose sensors, developing non-enzymatic glucose sensors are therefore becoming increasingly important as an alternative technique [5]. To design non-enzymatic alternatives with outstanding glucose catalytic efficiencies and stabilities, many efforts have been made. For example, noble metal platinum [6], palladium [7], and their alloys [8–10] have been widely utilized for non-enzymatic glucose detection. Nevertheless, the adsorption of intermediate/chloride compounds usually causes damage to the surface and reduces the electrocatalytic activity of noble metals [11]. Meanwhile, most active compounds coexisting in blood (for instance, ascorbic acid (AA), dopamine (DA), uric acid (UA), lactic acid (LA), and glutaric acid (GA)) will cause unavoidable interferences [12]. Finally, the high cost and scarcity of noble metals also limits their commercialization for glucose sensor applications.

To overcome the drawbacks of noble metals and the GO_x enzyme, developing active, stable, and inexpensive glucose sensors based on non-enzymatic compounds is still extremely urgent. Until now, large amounts of compounds based on transition metals (nickel (Ni), copper (Cu), cobalt (Co), etc.) or transition metal oxides (ferroferric oxide (Fe_3O_4), nickel oxide (NiO), copper oxide (CuO), etc.) have been widely researched and utilized for glucose sensors [5,11,13–17]. Among these non-enzymatic glucose sensors, Co-based oxides (CoO_x) are of interest due to their specific morphologies/structures, low-cost, polyvalence characteristics and excellent catalytic activities [18,19]. However, the weak electron conductivity of CoO_x is a significant drawback that prevents it from exhibiting excellent glucose detection properties [12]. Two strategies are usually adopted to promote the electron transport capacities of metal oxides: (1) Perfect electron conductors are introduced, for example, graphitic carbon matrices, as carbon matrices can promote the electron transport efficiencies of CoO_x in electrochemical reactions [20–22]. (2) Metallic particles coupled with graphitic carbon matrices are designed. For this kind of compound, in electrochemical catalysis processes, the outer surfaces of metallic particles transform into active sites with high valence states. For Co@CoO_x core-shell structures coupled with graphitic carbon matrices, in electrochemical catalysis processes, the inner

metallic core can enhance the transport of electrons derived from the outer active sites [23,24]. On the other hand, a deficiency of active sites is another drawback that limits the electrocatalytic abilities of metal oxides [25]. Introducing mesoporous carbon matrices with large surface areas and abundant carbon defects can obviously enhance the number of electrochemically available active sites [26,27]. This occurs because the abundant mesopores and carbon edges/defects can efficiently prevent the aggregation of active sites [28,29]. Based on the abovementioned facts, small Co nanoparticles embedded in porous carbons may have great potential for the commercialization and industrialization of non-enzymatic glucose sensors.

It is well known that biological tissues containing abundant nutrient transport channels are perfect precursors for constructing porous carbons. A paper towel is made from plant tissues with various porous channels. Herein, by strategically utilizing the porous structures of paper towels, cobalt acetates can uniformly adsorb and disperse in paper towels. After calcining in a nitrogen (N_2) atmosphere at 800°C , nitrogen-doped meso/macroporous carbon nanosheets embedded with small metallic Co nanoparticles (denoted as Co/NPCs) were successfully prepared. Several previous reports have revealed that combining a second transition metal element (such as Ni, Fe, or Mn) into Co-based materials is advantageous to improving the electrocatalytic efficiencies of the Co-based material [30–32]. Therefore, to enhance the intrinsic catalytic ability of the Co-based active sites, Fe has been chosen as the doping element. The as-prepared catalysts were characterized by using X-ray diffraction (XRD), scanning electron microscopy (SEM), transmission electron microscopy (TEM), N_2 adsorption-desorption, Raman spectroscopy, inductively coupled plasma-optical emission spectrometry (ICP-OES) and X-ray photoelectron spectroscopy (XPS) techniques. After optimizing the doping amounts of Fe atoms, the optimal $\text{Co}_7\text{Fe}_3/\text{NPCs}$ material exhibited the best glucose oxidation ability compared with the other control samples (such as $\text{Fe}_2\text{C-Co}_3\text{Fe}_7/\text{NPCs}$, CoFe/NPCs , and Co/NPCs). On the other hand, $\text{Co}_7\text{Fe}_3/\text{NPCs}$ modified electrodes are appropriate for detecting glucose concentrations in not only standard glucose samples but also real human serum samples.

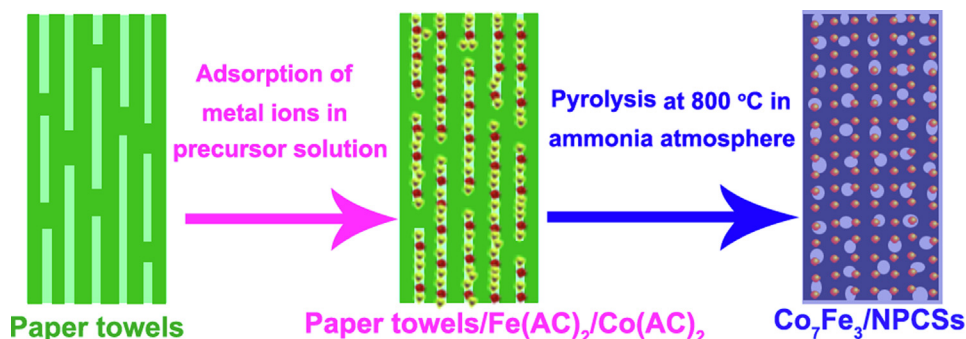
2. Experimental

2.1. Reagents and apparatus

Glucose, UA, DA, AA, LA, GA, cobalt acetate ($\text{Co}(\text{AC})_2 \cdot 4\text{H}_2\text{O}$), iron acetate ($\text{Fe}(\text{AC})_2 \cdot 4\text{H}_2\text{O}$), cyanamide (CN_2H_2), saccharose ($\text{C}_{12}\text{H}_{22}\text{O}_{11}$), sodium chloride (NaCl) and sodium hydroxide (NaOH) were purchased from Sinopharm Chemical Reagent Co. All of these chemical reagents were of analytical grade and used as received. The Fresh Air brand paper towels were purchased from Walmart. Ultrapure water (resistivity: $\rho \geq 18\text{ M}\Omega\text{ cm}^{-1}$) was utilized to prepare the solutions.

2.2. Catalyst synthesis

Scheme 1 exhibits the detailed method used to synthesize the $\text{Co}_7\text{Fe}_3/\text{NPCs}$. Approximately 30 mL of a solution containing 0.9 mmol $\text{Fe}(\text{AC})_2$ and 2.1 mmol $\text{Co}(\text{AC})_2$ was prepared and used as the impregnating solution (denoted as $[\text{Fe}(\text{AC})_2]_{0.9}\text{--}[\text{Co}(\text{AC})_2]_{2.1}$). Then, 1 g paper towels were impregnated for 24 h in



Scheme 1. Procedures used to synthesize the $\text{Co}_7\text{Fe}_3/\text{NPCSS}$ material.

the aforementioned $[\text{Fe}(\text{AC})_2]_{0.9}-[\text{Co}(\text{AC})_2]_{2.1}$ impregnating solution. Closely afterwards, the resultant paper towels- $[\text{Fe}(\text{AC})_2]_{0.9}-[\text{Co}(\text{AC})_2]_{2.1}$ solid substances were dried at 40°C and pyrolyzed in a N_2 atmosphere for 2 h at 800°C . For comparison, the $\text{Fe}_2\text{C}/\text{NPCSS}$, $\text{Fe}_2\text{C}-\text{Co}_3\text{Fe}_7/\text{NPCSS}$, CoFe/NPCSS and Co/NPCSS control materials were synthesized by utilizing $[\text{Fe}(\text{AC})_2]_{3.0}$, $[\text{Fe}(\text{AC})_2]_{2.1}-[\text{Co}(\text{AC})_2]_{0.9}$, $[\text{Fe}(\text{AC})_2]_{1.5}-[\text{Co}(\text{AC})_2]_{1.5}$ and $[\text{Co}(\text{AC})_2]_{0.3}$ impregnating solutions, respectively. All the other conditions used to synthesize the $\text{Fe}_2\text{C}/\text{NPCSS}$, $\text{Fe}_2\text{C}-\text{Co}_3\text{Fe}_7/\text{NPCSS}$, CoFe/NPCSS and Co/NPCSS were the same as those used to synthesize the $\text{Co}_7\text{Fe}_3/\text{NPCSS}$. On the other hand, to determine the critical roles of the porous structures in paper towels for boosting the catalysis efficiency of the $\text{Co}_7\text{Fe}_3/\text{NPCSS}$, we also prepared another control sample (denoted as $\text{Co}_7\text{Fe}_3/\text{NC}$) that has few porous structures. The precursor of the $\text{Co}_7\text{Fe}_3/\text{NC}$ is a mixture containing 0.9 mmol $\text{Fe}(\text{AC})_2$, 2.1 mmol $\text{Co}(\text{AC})_2$, 0.36 mmol CN_2H_2 and 2.9 mmol $\text{C}_{12}\text{O}_{11}\text{H}_{22}$.

2.3. Physical characterization of the resultant materials

The physical characterization techniques used in this paper (such as SEM, XRD, TEM, Raman, N_2 adsorption-desorption, ICP-OES, XPS and energy dispersive X-ray spectroscopy (EDS)) are detailed in the [supporting information \(SI\)](#).

2.4. Electrochemical tests

All the working electrodes used in this paper were glassy carbon electrodes (GCE, $d = 3$ mm) modified by the resultant catalysts. At first, the GCEs were polished by using 1.00, 0.30, and $0.05\ \mu\text{m}$ alumina powders. Afterwards, the polished GCEs were cleaned by using a HNO_3 (1:1) solution, ethanol, and deionized water. Various catalyst inks were prepared by mixing 3 mg catalyst powders in a 1 mL Nafion solution (0.1 wt%). After the ultrasonication treatment, $10\ \mu\text{L}$ as-prepared catalyst inks were then deposited onto the clean GCEs and dried under an infrared lamp. All the electrochemical measurements [cyclic voltammetry (CV) and chronoamperometry ($i-t$)] were performed with a CHI 830B electrochemical workstation (CH Instruments, USA) by using a traditional three-electrode configuration. The catalyst-modified GCEs, Ag/AgCl (in a saturated KCl solution), and platinum wire served as the working, reference, and counter electrodes, respectively. The current density (j) values displayed in this work were recorded as the ratios of the measured current values to the geometric area of the GCE. For the electrochemical tests, all the catalyst-modified GCEs were first analyzed with CV by scanning between 0.0 and $+0.65\ \text{V}$ vs. Ag/AgCl until the CV curves remained unchanged.

3. Results and discussion

3.1. Material characterization

The SEM images of the paper towel recorded at different magnifications are shown in [Fig. S1a–c](#), which reveals giant fiber textures with micron-sized diameters and rough surfaces. The amplifying SEM image further shows that the surfaces of the giant fibers disperse vast corrugated nanosheets ([Fig. S1d](#)). The N_2 adsorption-desorption isotherm of the paper towel ([Fig. S1e](#)) demonstrates a Brunauer-Emmett-Teller (BET) surface area of $16.39\ \text{m}^2\ \text{g}^{-1}$. In particular, the pore diameter distribution curve (inset of [Fig. S1e](#)) further reveals that the paper towel contains various porous structures with diameters between 2 and 200 nm. The structural characterization results prove the porosity of the paper towel precursor. The elemental mapping analysis of the paper towel precursor is carried out by utilizing EDX ([Fig. S2](#)). The elemental mapping image shown in [Fig. S2b](#) demonstrates that N- and O-based species are homogeneously distributed along with C atoms. The C ([Fig. S2c](#)), O ([Fig. S2d](#)), and N ([Fig. S2e](#)) concentrations of the paper towel are 53 at.%, 29 at.%, and 4 at.%, respectively. In particular, the O:C ratio is calculated to be ~ 0.547 .

After pyrolyzing in a N_2 atmosphere at 800°C , the various control samples were successfully synthesized. The XRD patterns ([Fig. 1](#)) of the $\text{Fe}_2\text{C}/\text{NPCSS}$, $\text{Fe}_2\text{C}-\text{Co}_3\text{Fe}_7/\text{NPCSS}$, CoFe/NPCSS , $\text{Co}_7\text{Fe}_3/\text{NPCSS}$, and Co/NPCSS samples all display (0 0 2) diffraction peaks, demonstrating the existence of graphitic carbon. Nevertheless, all the control sample XRD patterns have entirely different diffraction peaks corresponding to metallic phases. For instance, the XRD pattern of the $\text{Fe}_2\text{C}/\text{NPCSS}$ ([Fig. 1a](#)) only has characteristic peaks of the pure Fe_2C phase (PDF#36-1249) [33]. Nevertheless, the characteristic peaks of both Fe_2C (PDF#36-1249) and the Co_3Fe_7 alloy (PDF#48-1817) [34] simultaneously appear in the XRD pattern of the $\text{Fe}_2\text{C}-\text{Co}_3\text{Fe}_7/\text{NPCSS}$ ([Fig. 1b](#)). Even so, the peak intensities of Fe_2C are much weaker than those of the Co_3Fe_7 alloy. The XRD patterns of the CoFe/NPCSS ([Fig. 1c](#)), $\text{Co}_7\text{Fe}_3/\text{NPCSS}$ ([Fig. 1d](#)), and Co/NPCSS ([Fig. 1e](#)) have only diffraction peaks corresponding to the pure CoFe alloy (PDF#49-1568) [35], Co_7Fe_3 alloy (PDF#50-0795) [31], and Co (PDF#15-0806) [31], respectively. As seen in [Fig. 1b, c, d](#), the peak positions assigned to Co_7Fe_3 are quite similar to those of Co_3Fe_7 and CoFe . Therefore, we further compared the magnified XRD (1 1 0), (2 0 0) and (2 1 1) peaks of the Co_3Fe_7 , CoFe and Co_7Fe_3 alloys in [Fig. S3](#). It is clear that the (1 1 0), (2 0 0) and (2 1 1) peaks in the CoFe alloy XRD pattern occur at much higher diffraction angles than those in the Co_3Fe_7 alloy XRD pattern. Similarly, the (1 1 0), (2 0 0) and (2 1 1) peaks in the Co_7Fe_3 alloy XRD pattern occur at much higher diffraction angles than those in the CoFe alloy XRD pattern. These analysis results demonstrate that the alloy particles dispersed on the different control samples indeed have different chemical compositions.

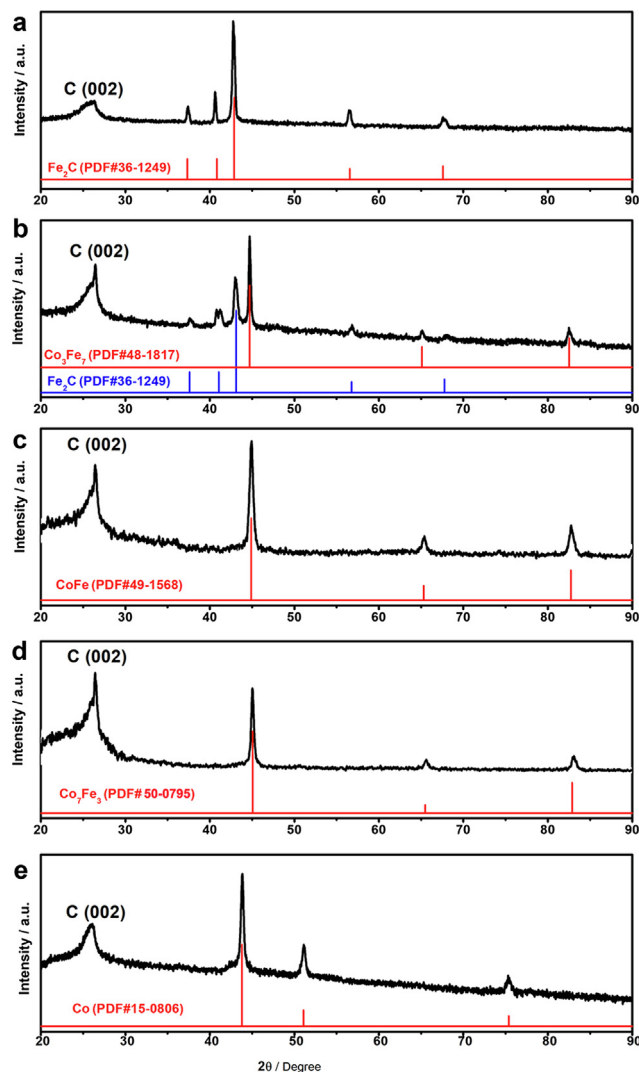


Fig. 1. XRD patterns of the (a) $\text{Fe}_2\text{C}/\text{NPCSS}$, (b) $\text{Fe}_2\text{C}-\text{Co}_3\text{Fe}_7/\text{NPCSS}$, (c) CoFe/NPCSS , (d) $\text{Co}_7\text{Fe}_3/\text{NPCSS}$, and (e) Co/NPCSS .

The SEM image shows that the $\text{Co}_7\text{Fe}_3/\text{NPCSS}$ consist of large ribbons with diameters larger than $10\ \mu\text{m}$ (Fig. S4a), and the surfaces of the large ribbons are covered by vast corrugated carbon nanosheets. Nevertheless, compared with the precursor ribbons shown in Fig. S1, parts of the giant fibers have decomposed into bulk particles in the pyrolysis processes. The magnified SEM image of the $\text{Co}_7\text{Fe}_3/\text{NPCSS}$ further demonstrates that the vast highly dispersed Co_7Fe_3 nanoparticles with diameters less than $50\ \text{nm}$ uniformly dispersed along the surfaces of the corrugated carbon nanosheets (Fig. S4b). To observe the surface structure characteristics of each material in detailed, the TEM images were shown in Fig. 2. Fig. 2 shows that the vast metallic nanoparticles have embedded into the frameworks of the porous carbon nanosheets for all the control samples (Fig. 2a–e). The statistical particle size results reveal that the average particle sizes are 35.18 , 33.99 , 32.87 , 30.71 , and $29.97\ \text{nm}$ for the $\text{Fe}_2\text{C}/\text{NPCSS}$ (Fig. S5a), $\text{Fe}_2\text{C}-\text{Co}_3\text{Fe}_7/\text{NPCSS}$ (Fig. S5b), CoFe/NPCSS (Fig. S5c), $\text{Co}_7\text{Fe}_3/\text{NPCSS}$ (Fig. 2f) and Co/NPCSS (Fig. S5d), respectively. The characterization results demonstrate that channels/pores existing in the plant tissues can promote the dispersion of metal salts and restrict the infinite growth of the alloy particles. On the other hand, the particle dispersion is better in the $\text{Co}_7\text{Fe}_3/\text{NPCSS}$ and Co/NPCSS than in the $\text{Fe}_2\text{C}/\text{NPCSS}$, $\text{Fe}_2\text{C}-\text{Co}_3\text{Fe}_7/\text{NPCSS}$ and CoFe/NPCSS . The magnified TEM

image of the $\text{Co}_7\text{Fe}_3/\text{NPCSS}$ clearly shows that vast mesopores with different diameters have formed along the surfaces of the $\text{Co}_7\text{Fe}_3/\text{NPCSS}$ (Fig. 2g). In addition, Fig. 2h displays the high resolution transmission electron microscopy (HRTEM) image obtained from an individual nanocrystal, and the well-resolved lattice fringes with interplanar spacings of $0.201\ \text{nm}$ correspond to the $(1\ 1\ 0)$ planes of the Co_7Fe_3 crystal.

The porosities of all the control samples were further characterized by a N_2 adsorption-desorption technique. As shown in Fig. 3a, the N_2 adsorption-desorption isotherms of the five control samples are extremely similar to one another and exhibit characteristics of type IV adsorption isotherms, indicating the wide distribution of porous structures. The specific surface areas calculated from the adsorption data by using the BET method are shown in Fig. 3b, in which all the control samples exhibit similar BET surface areas: $361.9\ \text{m}^2\ \text{g}^{-1}$ for the $\text{Fe}_2\text{C}/\text{NPCSS}$, $358.3\ \text{m}^2\ \text{g}^{-1}$ for the $\text{Fe}_2\text{C}-\text{Co}_3\text{Fe}_7/\text{NPCSS}$, $355.1\ \text{m}^2\ \text{g}^{-1}$ for the CoFe/NPCSS , $347.9\ \text{m}^2\ \text{g}^{-1}$ for the $\text{Co}_7\text{Fe}_3/\text{NPCSS}$, and $344.5\ \text{m}^2\ \text{g}^{-1}$ the Co/NPCSS . The pore diameter dispersion curves are similar for the five control samples (Fig. 3c). Nevertheless, compared with the pore diameter dispersion curve of the paper towel (shown in the inset of Fig. S1d), more mesopores with diameters smaller than $10\ \text{nm}$ appear in all the samples after the pyrolysis processes. Previous reports have proved that these kinds of mesopores are derived from the release of $\text{H}_2\text{O}/\text{CO}_2$ gases during the carbonation process [31]. Specifically, the amount of mesopores with sizes between 10 and $50\ \text{nm}$ for the $\text{Fe}_2\text{C}/\text{NPCSS}$ and $\text{Fe}_2\text{C}-\text{Co}_3\text{Fe}_7/\text{NPCSS}$ is much larger than that of the CoFe/NPCSS , $\text{Co}_7\text{Fe}_3/\text{NPCSS}$ and Co/NPCSS . This abnormal phenomenon can be attributed to the distribution of metallic nanoparticles on the porous carbon sheets. The HRTEM image (Fig. 2g) proves that the small alloy nanoparticles with diameters between 10 and $50\ \text{nm}$ are mainly embedded in the mesopores of the carbon matrices. As shown in Figs. 2 and S5, the CoFe/NPCSS , $\text{Co}_7\text{Fe}_3/\text{NPCSS}$, and Co/NPCSS contain more alloy nanoparticles (with diameters between 10 and $50\ \text{nm}$) than the $\text{Fe}_2\text{C}/\text{NPCSS}$ and $\text{Fe}_2\text{C}-\text{Co}_3\text{Fe}_7/\text{NPCSS}$. Therefore, most of the mesopores (with sizes between 10 and $50\ \text{nm}$) of the CoFe/NPCSS , $\text{Co}_7\text{Fe}_3/\text{NPCSS}$, and Co/NPCSS are filled with alloy nanoparticles with diameters between 10 and $50\ \text{nm}$, leading to fewer mesopores with sizes between 10 and $50\ \text{nm}$ in the pore diameter dispersion curves (Fig. 3c). The Raman spectrum of the $\text{Co}_7\text{Fe}_3/\text{NPCSS}$ (Fig. 3d) shows four broad and overlapping peaks. Among the four peaks, the D ($1351\ \text{cm}^{-1}$) and G ($1587\ \text{cm}^{-1}$) peaks are fairly sharp, indicating the coexisting graphitic structures and carbon defects. An I_D/I_G value of 1.33 proves the existence of abundant carbon defects/edges.

To further investigate the elemental composition of the $\text{Co}_7\text{Fe}_3/\text{NPCSS}$ catalyst, the Co and Fe contents were detected by an ICP-OES technique (3 replicate experiments were performed). The measurement results show that the mole numbers of Co and Fe atoms in $1\ \text{mg}$ $\text{Co}_7\text{Fe}_3/\text{NPCSS}$ are $1.0358 \times 10^{-3}\ \text{mmol}$ and $2.4067 \times 10^{-3}\ \text{mmol}$ respectively, demonstrating a Co/Fe molar ratio of $7:3.0127$. This molar ratio value is nearly consistent with the $7:3$ ratio of the Co_7Fe_3 alloy determined from the XRD pattern of the $\text{Co}_7\text{Fe}_3/\text{NPCSS}$. To further study the chemical states of the metallic alloy and carbon matrices, XPS was utilized. The XPS survey spectrum confirms the presences of C, O, N, Fe, and Co elements in the $\text{Co}_7\text{Fe}_3/\text{NPCSS}$ (Fig. 4a). Fig. 4b shows the high-resolution C $1s$ XPS spectrum of the $\text{Co}_7\text{Fe}_3/\text{NPCSS}$, and the characteristic peaks centered at binding energies of 284.7 , 285.6 , and $288.4\ \text{eV}$ can be assigned to the C–C sp^2 , C–N $\text{sp}^2/\text{C}=\text{C}\ \text{sp}^3$, and C–O groups, respectively. The high resolution O $1s$ XPS spectrum of the $\text{Co}_7\text{Fe}_3/\text{NPCSS}$ (Fig. 4c) shows two sharp peaks located at ~ 532.2 and $\sim 533.8\ \text{eV}$, which are assigned to the oxygen-containing groups dispersed on the material surfaces. In addition, another fairly weak peak located at $\sim 529.8\ \text{eV}$ can be observed in the high resolution O $1s$ XPS

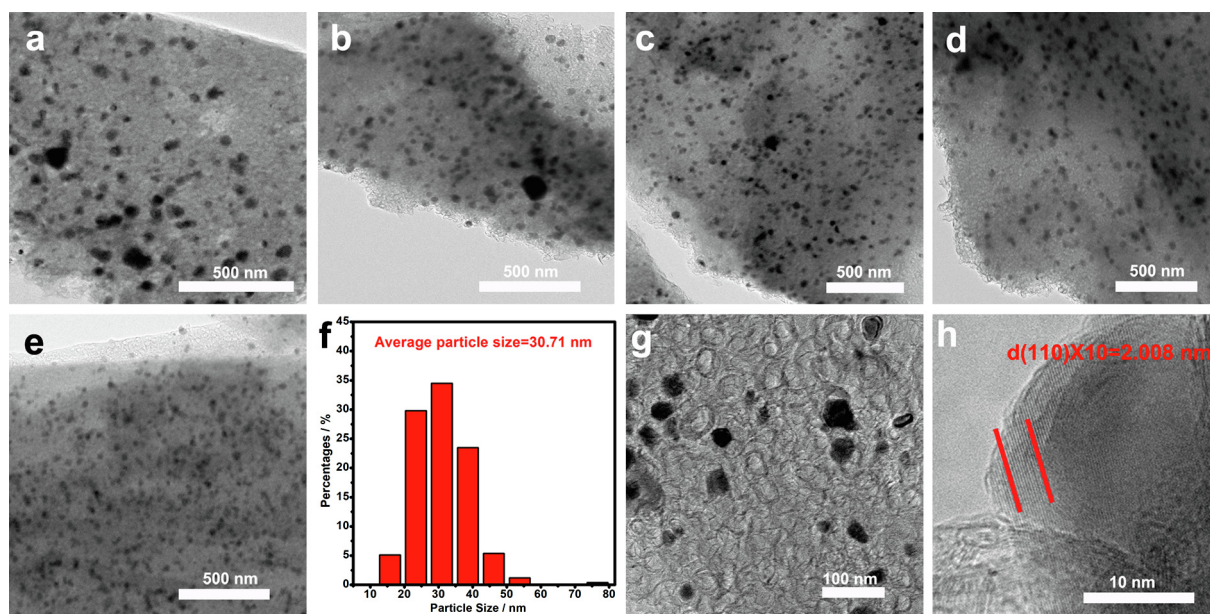


Fig. 2. TEM images of the $\text{Fe}_2\text{C}/\text{NPCSS}$ (a), $\text{Fe}_2\text{C}-\text{Co}_3\text{Fe}_7/\text{NPCSS}$ (b), CoFe/NPCSS (c), $\text{Co}_7\text{Fe}_3/\text{NPCSS}$ (d) and Co/NPCSS (e). (f) Particle size distribution curve of the $\text{Co}_7\text{Fe}_3/\text{NPCSS}$. (g, h) HRTEM images of the $\text{Co}_7\text{Fe}_3/\text{NPCSS}$.

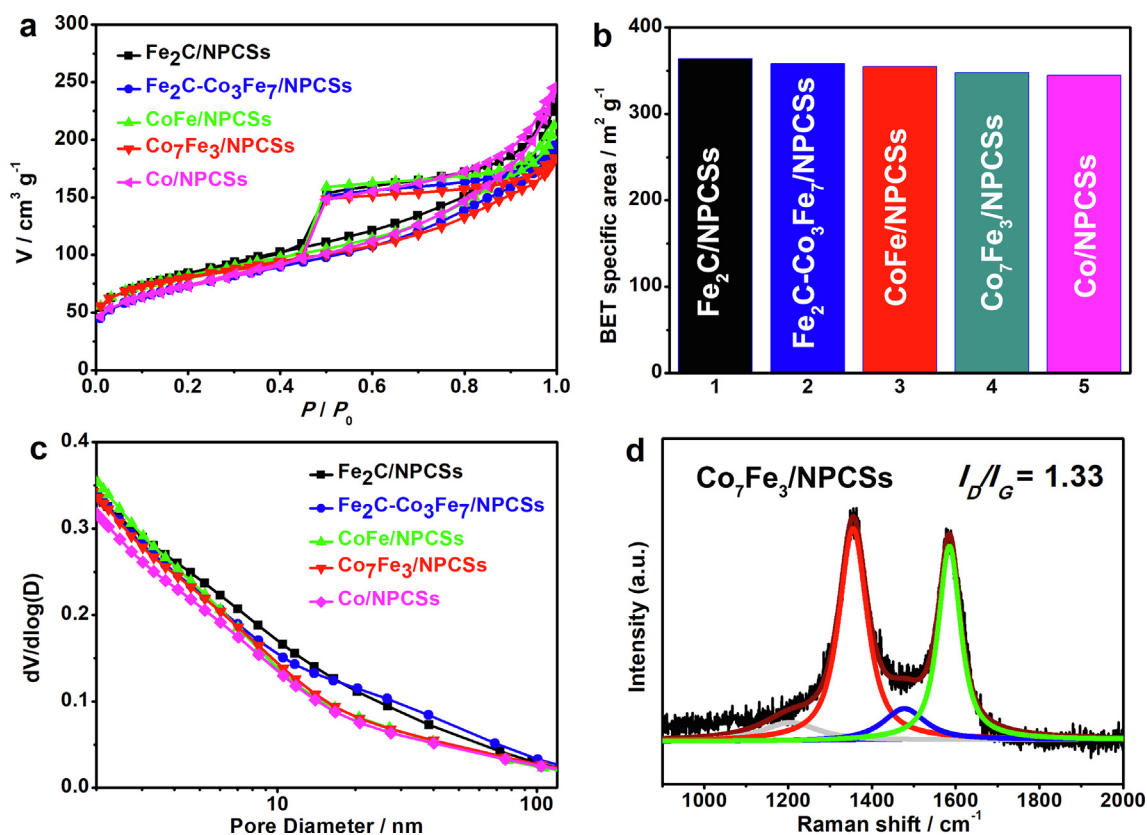


Fig. 3. (a) N_2 adsorption-desorption isotherms, (b) histograms of the BET specific areas and (c) pore diameter distribution curves for the $\text{Fe}_2\text{C}/\text{NPCSS}$, $\text{Fe}_2\text{C}-\text{Co}_3\text{Fe}_7/\text{NPCSS}$, CoFe/NPCSS , $\text{Co}_7\text{Fe}_3/\text{NPCSS}$ and Co/NPCSS control samples. (d) Raman spectrum of the $\text{Co}_7\text{Fe}_3/\text{NPCSS}$ sample.

spectrum of the $\text{Co}_7\text{Fe}_3/\text{NPCSS}$, which is due to the slight oxidation of the alloy particles on the surfaces. Even so, due to the trace metal oxides, the crystallographic planes were not detected in the XRD pattern (Fig. 1d). As shown in Fig. 4d, the high-resolution N 1s XPS spectrum shows that the N atoms doped into the carbon structures exist in the form of pyridinic N (~ 398.4 eV), pyrrolic N

(~ 400.7 eV), graphitic N (~ 402.4 eV), and oxidized N (~ 406.5 eV), respectively [36]. Finally, to confirm the chemical states of the Co_7Fe_3 alloy nanoparticles, we have fitted the high resolution Co $2p_{3/2}$ (Fig. 4e) and Fe $2p_{3/2}$ (Fig. 4f) XPS spectra of the $\text{Co}_7\text{Fe}_3/\text{NPCSS}$. It is clear that both the high resolution Co $2p_{3/2}$ (Fig. 4e) and Fe $2p_{3/2}$ (Fig. 4f) XPS spectra exhibit obvious peaks corresponding

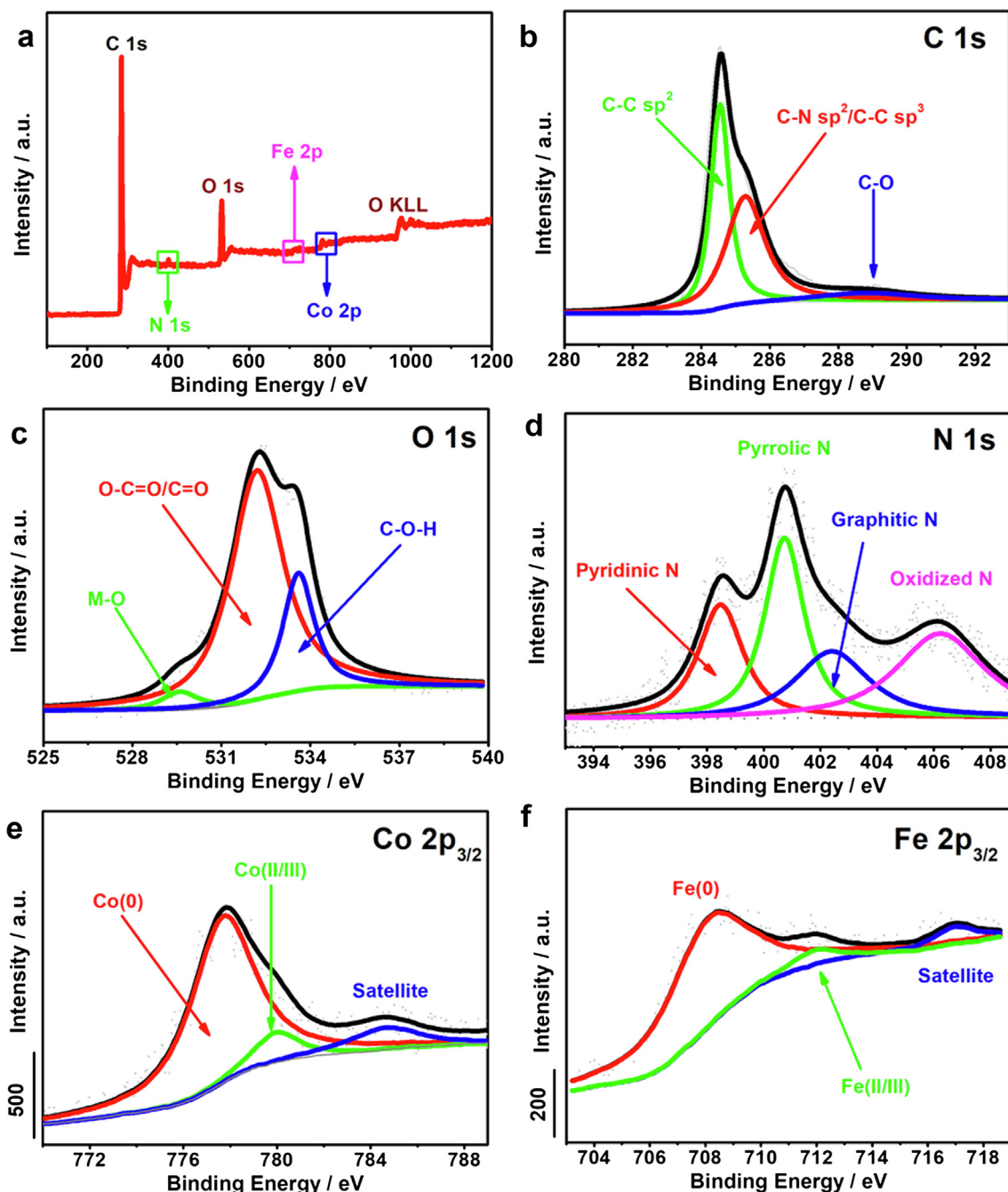


Fig. 4. The XPS survey spectrum (a) and high-resolution XPS spectra of C 1s (b), O 1s (c), N 1s (d), Co 2p (e), and Fe 2p (f) for the $\text{Co}_7\text{Fe}_3/\text{NPCSS}$.

to metallic Co (~ 777.9 eV) and metallic Fe (~ 707.9 eV), demonstrating the successful formation of the Co_7Fe_3 alloy. The weak characteristic peaks of the metal oxides (where the Co oxides correspond to ~ 780.2 eV, and the Fe oxides correspond to ~ 711.5 eV) and satellites appear in the high resolution $\text{Co } 2p_{3/2}$ and $\text{Fe } 2p_{3/2}$ XPS spectra, indicating the slight oxidation of the Co_7Fe_3 alloy nanoparticles' surfaces. Nevertheless, the weak characteristic peaks indicate the minor oxidation of the Co_7Fe_3 alloy.

3.2. The electrochemical oxidation of glucose on different catalyst-modified electrodes

A comparative study of the electrocatalytic performance toward glucose oxidation was implemented for the five control catalyst-modified GCEs in a 0.1 M NaOH solution. The CV curves measured

in the absence (black lines) and presence (red lines) of 1.0 mM glucose are shown in Fig. 5a–e. No oxidation peaks were present in the CV curves of the $\text{Fe}_2\text{C}/\text{NPCSS}/\text{Nafion}/\text{GCE}$ (Fig. 5a). By adding a 1 mM glucose solution to the electrolyte, only weak current responses ($\Delta j = j_{\text{with 1.0 mM glucose}} - j_{\text{without glucose}}$) can be found in the selected potential window. The CV curves of the $\text{Fe}_2\text{C}-\text{Co}_3\text{Fe}_7/\text{NPCSS}/\text{Nafion}/\text{GCE}$ (Fig. 5b) have distinct different shapes compared with those of the $\text{Fe}_2\text{C}/\text{NPCSS}/\text{Nafion}/\text{GCE}$ (Fig. 5a). In the absence of the 1.0 mM glucose solution, an obvious redox pair attributed to the irreversible redox processes of $\text{Co}^{2+}/\text{Co}^{3+}$ appears in the potential region of +0.10 to +0.45 V vs. Ag/AgCl for the $\text{Fe}_2\text{C}-\text{Co}_3\text{Fe}_7/\text{NPCSS}/\text{Nafion}/\text{GCE}$. The j values increase for increasing potentials from +0.45 to +0.65 V vs. Ag/AgCl in the positive scan, which is mainly attributed to the oxidation of the metal ions from Co^{3+} to Co^{4+} . Furthermore, once the 1.0 mM glucose solution was

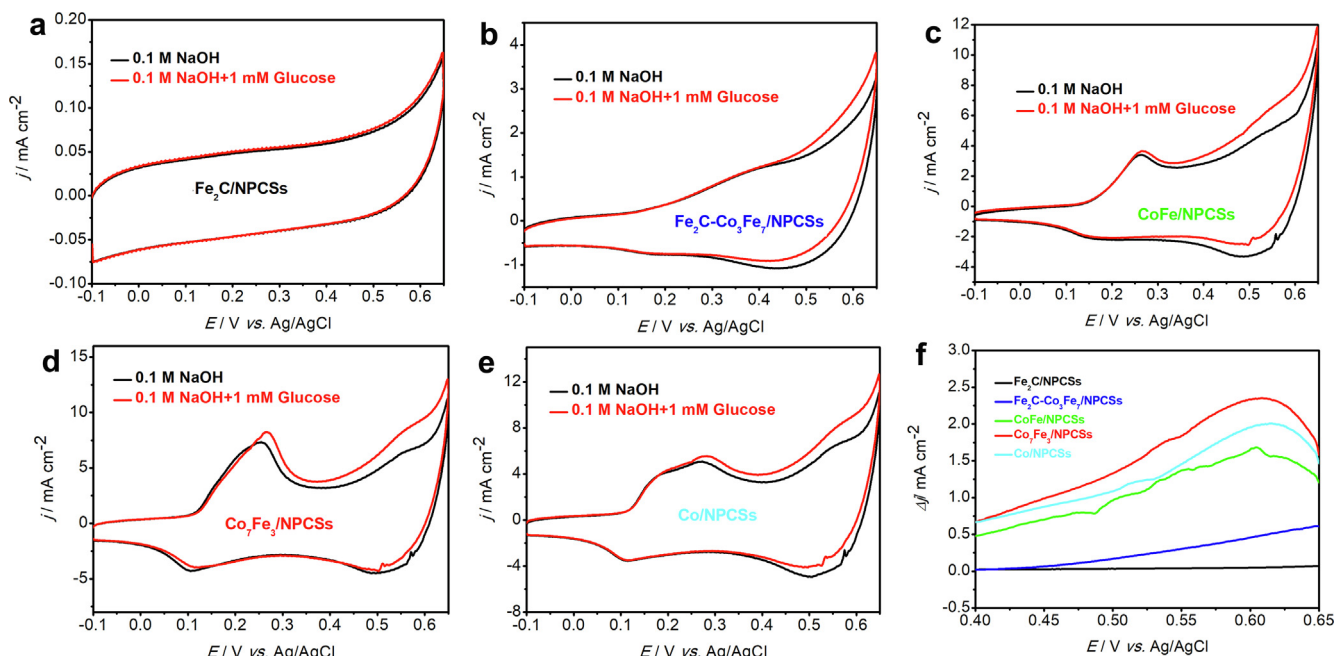


Fig. 5. CV plots of the (a) $\text{Fe}_2\text{C}/\text{NPCSSs}/\text{Nafion}/\text{GCE}$, (b) $\text{Fe}_2\text{C}-\text{Co}_3\text{Fe}_7/\text{NPCSSs}/\text{Nafion}/\text{GCE}$, (c) $\text{CoFe}/\text{NPCSSs}/\text{Nafion}/\text{GCE}$, (d) $\text{Co}_7\text{Fe}_3/\text{NPCSSs}/\text{Nafion}/\text{GCE}$, and (e) $\text{Co}/\text{NPCSSs}/\text{Nafion}/\text{GCE}$ measured in a 0.1 M NaOH solution prepared without (black lines) and with (red lines) 1.0 mM glucose (scan rate: 50 mV s^{-1}). (f) The corresponding Δj responses of the different catalyst-modified GCEs toward glucose oxidation in the potential range of +0.40 to +0.65 V vs. Ag/AgCl. (For interpretation of the references to colour in this figure legend, the reader is referred to the web version of this article.)

added to the 0.1 M NaOH solution, in the potential range of $E > +0.45 \text{ V vs. Ag/AgCl}$, the Δj values of glucose electro-oxidation increase gradually. The electrochemical oxidation of glucose corresponds to the formation of Co^{4+} ions, which proves that the high valence state Co^{4+} ions act as active sites for glucose oxidation. The CV curves of the $\text{CoFe}/\text{NPCSSs}/\text{Nafion}/\text{GCE}$ (Fig. 5c), $\text{Co}_7\text{Fe}_3/\text{NPCSSs}/\text{Nafion}/\text{GCE}$ (Fig. 5d), and $\text{Co}/\text{NPCSSs}/\text{Nafion}/\text{GCE}$ (Fig. 5e) reveal a similar trend. For instance, the irreversible redox pair of $\text{Co}^{2+}/\text{Co}^{3+}$ appears to be located in the potential region of +0.10 to +0.40 V vs. Ag/AgCl. When the working potentials are more positive than +0.40 V vs. Ag/AgCl, obvious oxidation peaks appeared at +0.55 V vs. Ag/AgCl. In particular, the Δj values toward glucose electro-oxidation appeared once the potential values were larger than +0.25 V vs. Ag/AgCl. Even though the CV curve shapes of the $\text{CoFe}/\text{NPCSSs}/\text{Nafion}/\text{GCE}$, $\text{Co}_7\text{Fe}_3/\text{NPCSSs}/\text{Nafion}/\text{GCE}$, and $\text{Co}/\text{NPCSSs}/\text{Nafion}/\text{GCE}$ are similar, their j values are very different. By comparing the glucose oxidation catalysis activities of the different control samples, the Δj values of glucose oxidation between +0.40 and +0.65 V vs. Ag/AgCl are calculated and shown in Fig. 5f. It is clear that in the as-selected potential range of +0.40 to +0.65 V vs. Ag/AgCl, the Δj values increase by the order of $\text{Fe}_2\text{C}/\text{NPCSSs}/\text{Nafion}/\text{GCE} < \text{Fe}_2\text{C}-\text{Co}_3\text{Fe}_7/\text{NPCSSs}/\text{Nafion}/\text{GCE} < \text{CoFe}/\text{NPCSSs}/\text{Nafion}/\text{GCE} < \text{Co}/\text{NPCSSs}/\text{Nafion}/\text{GCE} < \text{Co}_7\text{Fe}_3/\text{NPCSSs}/\text{Nafion}/\text{GCE}$. The $\text{Co}_7\text{Fe}_3/\text{NPCSSs}$ exhibit the best catalysis activity toward glucose oxidation. On the other hand, a potential value of +0.60 V vs. Ag/AgCl is the optimal glucose oxidation potential for the $\text{CoFe}/\text{NPCSSs}/\text{Nafion}/\text{GCE}$, $\text{Co}_7\text{Fe}_3/\text{NPCSSs}/\text{Nafion}/\text{GCE}$, and $\text{Co}/\text{NPCSSs}/\text{Nafion}/\text{GCE}$.

Fig. 6a shows the Nyquist plots obtained by electrochemical impedance spectroscopy (EIS) for the various Co-based control samples. It is clear that in the high frequency range, semicircles are not observed for any of the control samples. This result proves that embedding alloy nanoparticles into the graphitic carbon nanosheets indeed enhances the electron conductivity of each hybrid catalyst. The EIS tests show that the conductivity is not the primary factor leading to the differences in the catalysis

activities of the various control catalysts. The double-layer capacitances (C_{dl}) of the four control samples were extracted by plotting the $\Delta j_{\text{at } 0.55 \text{ V}/2}$ values [$\Delta j_{\text{at } 0.55 \text{ V}/2} = (j_a - j_c)/2$] against the CV scan rates (Fig. 6b). The j_a and j_c values are the corresponding oxidation and reduction j values recorded at +0.55 V vs. Ag/AgCl (as shown in Fig. S6). Fig. 6b shows that the C_{dl} values increase by the order of 10.67 mF cm^{-2} ($\text{Fe}_2\text{C}-\text{Co}_3\text{Fe}_7/\text{NPCSSs}$) $< 48.95 \text{ mF cm}^{-2}$ ($\text{CoFe}/\text{NPCSSs}$) $< 54.71 \text{ mF cm}^{-2}$ (Co/NPCSSs) $< 69.63 \text{ mF cm}^{-2}$ ($\text{Co}_7\text{Fe}_3/\text{NPCSSs}$). It is well known that the C_{dl} values are directly proportional to the electrochemically active surface areas, and thus, the electrochemically active surface areas also increase by the order of $\text{Fe}_2\text{C}-\text{Co}_3\text{Fe}_7/\text{NPCSSs} < \text{CoFe}/\text{NPCSSs} < \text{Co}/\text{NPCSSs} < \text{Co}_7\text{Fe}_3/\text{NPCSSs}$. Finally, Fig. 6c compares the CV curves of the $\text{Fe}_2\text{C}-\text{Co}_3\text{Fe}_7/\text{NPCSSs}/\text{Nafion}/\text{GCE}$, $\text{CoFe}/\text{NPCSSs}/\text{Nafion}/\text{GCE}$, $\text{Co}_7\text{Fe}_3/\text{NPCSSs}/\text{Nafion}/\text{GCE}$ and $\text{Co}/\text{NPCSSs}/\text{Nafion}/\text{GCE}$ measured in a 0.1 M NaOH solution prepared without glucose. It is clear that in the high potential range ($E > +0.45 \text{ V vs. Ag/AgCl}$), the j values recorded in the positive scan increase by the order of $\text{Fe}_2\text{C}-\text{Co}_3\text{Fe}_7/\text{NPCSSs}/\text{Nafion}/\text{GCE} < \text{CoFe}/\text{NPCSSs}/\text{Nafion}/\text{GCE} < \text{Co}/\text{NPCSSs}/\text{Nafion}/\text{GCE} < \text{Co}_7\text{Fe}_3/\text{NPCSSs}/\text{Nafion}/\text{GCE}$. In fact, in the high potential range of +0.45 to +0.65 V vs. Ag/AgCl, the oxidation j values correspond to the formation of high valence state Co^{4+} ions. After the electrochemical polarization, the amount of Co^{4+} active sites formed along the control catalysts' surfaces increased by the order of $\text{Fe}_2\text{C}-\text{Co}_3\text{Fe}_7/\text{NPCSSs} < \text{CoFe}/\text{NPCSSs} < \text{Co}/\text{NPCSSs} < \text{Co}_7\text{Fe}_3/\text{NPCSSs}$.

Combining the abovementioned structural, chemical and electrochemical characterization results, we have determined why the glucose oxidation performances of the $\text{Co}_7\text{Fe}_3/\text{NPCSSs}$ have improved significantly. As shown in Figs. 2a-f, S5, 3b and 6a, the morphologies, particle sizes, BET surface areas and electron conductivities are similar for all the control samples. Therefore, these properties are not the main factors leading to the large differences in the glucose oxidation activities. The large differences in the glucose oxidation activities of the various control catalysts containing different Co amounts are mainly attributed to the different surface chemical compositions of the control catalysts. The $\text{Fe}_2\text{C}/\text{NPCSSs}$

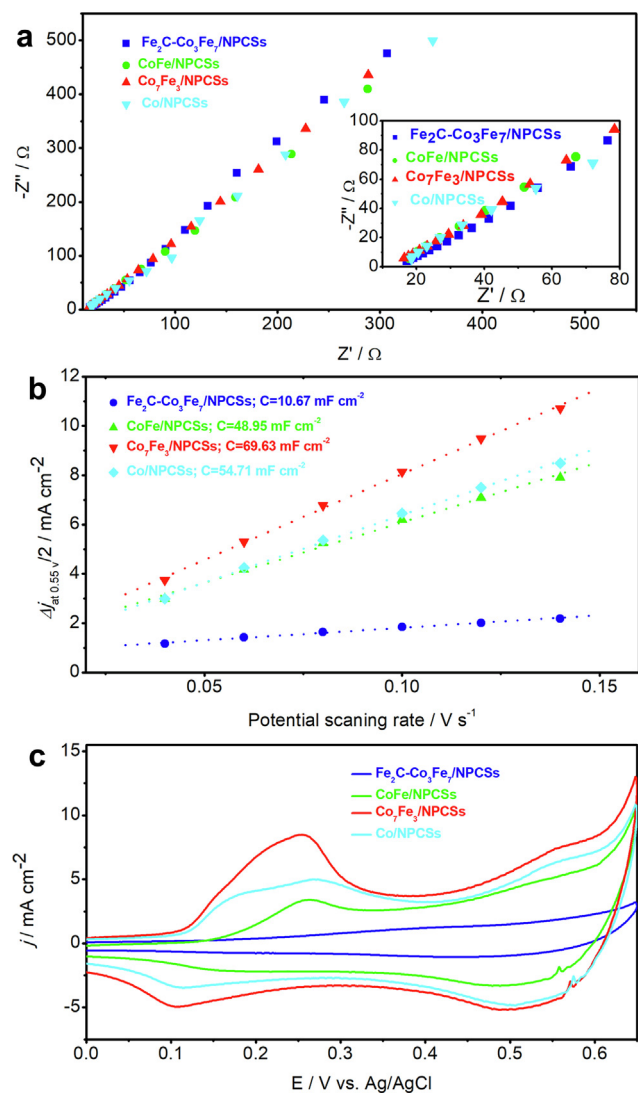


Fig. 6. (a) Nyquist plots obtained by EIS for the $\text{Fe}_2\text{C-Co}_3\text{Fe}_7/\text{NPCSSs}/\text{Nafion}/\text{GCE}$, $\text{CoFe}/\text{NPCSSs}/\text{Nafion}/\text{GCE}$, $\text{Co}_7\text{Fe}_3/\text{NPCSSs}/\text{Nafion}/\text{GCE}$, and $\text{Co}/\text{NPCSSs}/\text{Nafion}/\text{GCE}$. Inset shows the magnified images in the high frequency region. (b) The capacitive current responses recorded at +0.55 V vs. Ag/AgCl ($\Delta j_{\text{at } +0.55 \text{ V vs. Ag/AgCl}}$) as a function of the scan rates for the different control samples. (c) A comparison of the CV plots of the $\text{Fe}_2\text{C-Co}_3\text{Fe}_7/\text{NPCSSs}/\text{Nafion}/\text{GCE}$, $\text{CoFe}/\text{NPCSSs}/\text{Nafion}/\text{GCE}$, $\text{Co}_7\text{Fe}_3/\text{NPCSSs}/\text{Nafion}/\text{GCE}$, and $\text{Co}/\text{NPCSSs}/\text{Nafion}/\text{GCE}$ measured in a 0.1 M NaOH solution with a scan rate of 50 mV s^{-1} .

without Co have an inferior glucose oxidation activity compared to the other catalysts, demonstrating that the active sites for glucose oxidation are Co-based species. Because the $\text{Fe}_2\text{C-Co}_3\text{Fe}_7/\text{NPCSSs}$, $\text{CoFe}/\text{NPCSSs}$ and $\text{Co}_7\text{Fe}_3/\text{NPCSSs}$ catalysts contain both Co and Fe elements, their C_{dl} values (Fig. 6b) and oxidation j values recorded from +0.45 to +0.65 V vs. Ag/AgCl in the positive scan of the CV curves increase by the order of $\text{Fe}_2\text{C-Co}_3\text{Fe}_7/\text{NPCSSs} < \text{CoFe}/\text{NPCSSs} < \text{Co}_7\text{Fe}_3/\text{NPCSSs}$ (Fig. 6c). The experimental results indicate that the many Co atoms in the $\text{Co}_7\text{Fe}_3/\text{NPCSSs}$ indeed causes more Co^{4+} active sites to form along the surfaces during the polarization process, which results in $\text{Co}_7\text{Fe}_3/\text{NPCSSs}$ having a larger electrochemically active surface area and better glucose oxidation ability than the $\text{Fe}_2\text{C-Co}_3\text{Fe}_7/\text{NPCSSs}$ and $\text{CoFe}/\text{NPCSSs}$. Finally, although the Co/NPCSSs have a higher Co content than the $\text{Co}_7\text{Fe}_3/\text{NPCSSs}$, the C_{dl} value (Fig. 6b) and oxidation j values recorded from +0.45 to +0.65 V vs. Ag/AgCl in the positive scan (Fig. 6c) for the Co/NPCSSs are lower than those of the $\text{Co}_7\text{Fe}_3/\text{NPCSSs}$. This reveals that the $\text{Co}_7\text{Fe}_3/\text{NPCSSs}$ catalyst contains more Co^{4+} active sites along the

surfaces after the polarization process, which results in the $\text{Co}_7\text{Fe}_3/\text{NPCSSs}$ catalyst having a larger electrochemically active surface area than the Co/NPCSSs . Therefore, the $\text{Co}_7\text{Fe}_3/\text{NPCSSs}$ also have a better catalysis activity than the Co/NPCSSs toward glucose oxidation. A small number of Fe atoms in the Co_7Fe_3 alloy can cause more Co^{4+} active sites to form along the catalysts' surfaces, which is similar to conclusions reported previously in the literature [32]. The HRTEM image (shown in Fig. S7) shows that after carrying out the electrochemical polarization, thin shell layers (with thicknesses of $\sim 5 \text{ nm}$) can be clearly observed surrounding the surfaces of all the Co_7Fe_3 nanoparticles. This phenomenon proves that metallic oxide layers with high metal valences have indeed formed, which is similar to conclusions reported previously in the literature [28].

On the other hand, to determine the effect of pore effect of the plant-tissue-based $\text{Co}_7\text{Fe}_3/\text{NPCSSs}$, we have also synthesized a $\text{Co}_7\text{Fe}_3/\text{NC}$ control sample by utilizing a mixed precursor containing 0.9 mmol $\text{Fe}(\text{AC})_2$, 2.1 mmol $\text{Co}(\text{AC})_2$, 0.36 mmol CN_2H_2 and 2.9 mmol $\text{C}_{12}\text{O}_{11}\text{H}_{22}$. As shown in the SEM image (Fig. S8a), the $\text{Co}_7\text{Fe}_3/\text{NC}$ consists of a vast bulk material with poor porosity. The TEM image (Fig. S8b) exhibits that the surfaces of the $\text{Co}_7\text{Fe}_3/\text{NC}$ contain few porous structures, and the alloy nanoparticles are thoroughly embedded in the carbon matrices. As shown in Fig. S9a, the CV curve of the $\text{Co}_7\text{Fe}_3/\text{NC}/\text{Nafion}/\text{GCE}$ measured in the 0.1 M NaOH solution without glucose is much weaker than that of the $\text{Co}_7\text{Fe}_3/\text{NPCSSs}/\text{Nafion}/\text{GCE}$. By adding a 1.0 mM glucose solution to a 0.1 M NaOH solution, the $\text{Co}_7\text{Fe}_3/\text{NC}/\text{Nafion}/\text{GCE}$ also exhibits much smaller Δj values than the $\text{Co}_7\text{Fe}_3/\text{NPCSSs}/\text{Nafion}/\text{GCE}$ (Fig. S9b), indicating the improved catalysis ability of the $\text{Co}_7\text{Fe}_3/\text{NPCSSs}$ toward glucose oxidation compared with the $\text{Co}_7\text{Fe}_3/\text{NC}$. This control experiment obviously proves that the perfect porosity of the $\text{Co}_7\text{Fe}_3/\text{NPCSSs}$ indeed enhances the glucose catalytic activity.

From the abovementioned discussions, the superior catalysis ability of the $\text{Co}_7\text{Fe}_3/\text{NPCSSs}$ toward glucose oxidation is mainly attributed to the following factors: (1) The superior electron conductivity of the $\text{Co}_7\text{Fe}_3/\text{NPCSSs}$ enhances the electron transfer rate of the electrochemical reactions. (2) The hierarchical porous channels derived from the plant tissues and the releases of $\text{H}_2\text{O}/\text{CO}_2$ gases in the pyrolysis process can promote mass transport during the electrochemical reaction. (3) The abundant meso/macropores afforded the catalyst with a large BET surface area and vast carbon edges/defects, improving the dispersion of the ultrafine alloy nanoparticles and increasing the number of active sites for glucose oxidation catalysis. (4) Doping small amount of Fe atoms into the Co nanoparticles can cause more high-valence state Co^{4+} active sites to form along the catalyst surfaces during the electrochemical polarization processes.

3.3. Amperometric performances of the different catalyst-modified GCEs toward glucose detection

We further measured the glucose sensor performances of the $\text{Fe}_2\text{C-Co}_3\text{Fe}_7/\text{NPCSSs}/\text{Nafion}/\text{GCE}$, $\text{CoFe}/\text{NPCSSs}/\text{Nafion}/\text{GCE}$, $\text{Co}_7\text{Fe}_3/\text{NPCSSs}/\text{Nafion}/\text{GCE}$ and $\text{Co}/\text{NPCSSs}/\text{Nafion}/\text{GCE}$ at the optimal potential of +0.60 V vs. Ag/AgCl by utilizing a chronoamperometry technique. The amperometric responses for successive addition of glucose at +0.60 V vs. Ag/AgCl were measured and are shown in Figs. 7a and S9. The response times (*i.e.*, the times required to reach 95% of the steady-state j values) increase by the order of $\text{Fe}_2\text{C-Co}_3\text{Fe}_7/\text{NPCSSs}/\text{Nafion}/\text{GCE}$ (4.0 s) < $\text{CoFe}/\text{NPCSSs}/\text{Nafion}/\text{GCE}$ (3.5 s) < $\text{Co}/\text{NPCSSs}/\text{Nafion}/\text{GCE}$ (2.8 s) < $\text{Co}_7\text{Fe}_3/\text{NPCSSs}/\text{Nafion}/\text{GCE}$ (2.2 s), demonstrating that $\text{Co}_7\text{Fe}_3/\text{NPCSSs}/\text{Nafion}/\text{GCE}$ has the shortest response time (Figs. 7b and S10). The calibration curves of the four control electrodes were plotted and can be divided into two linear ranges. In the low concentration ranges (Figs. 7c and S11), the detection limit values ($S/N = 3$) of the $\text{Fe}_2\text{C-Co}_3\text{Fe}_7/\text{NPCSSs}/\text{Nafion}/$

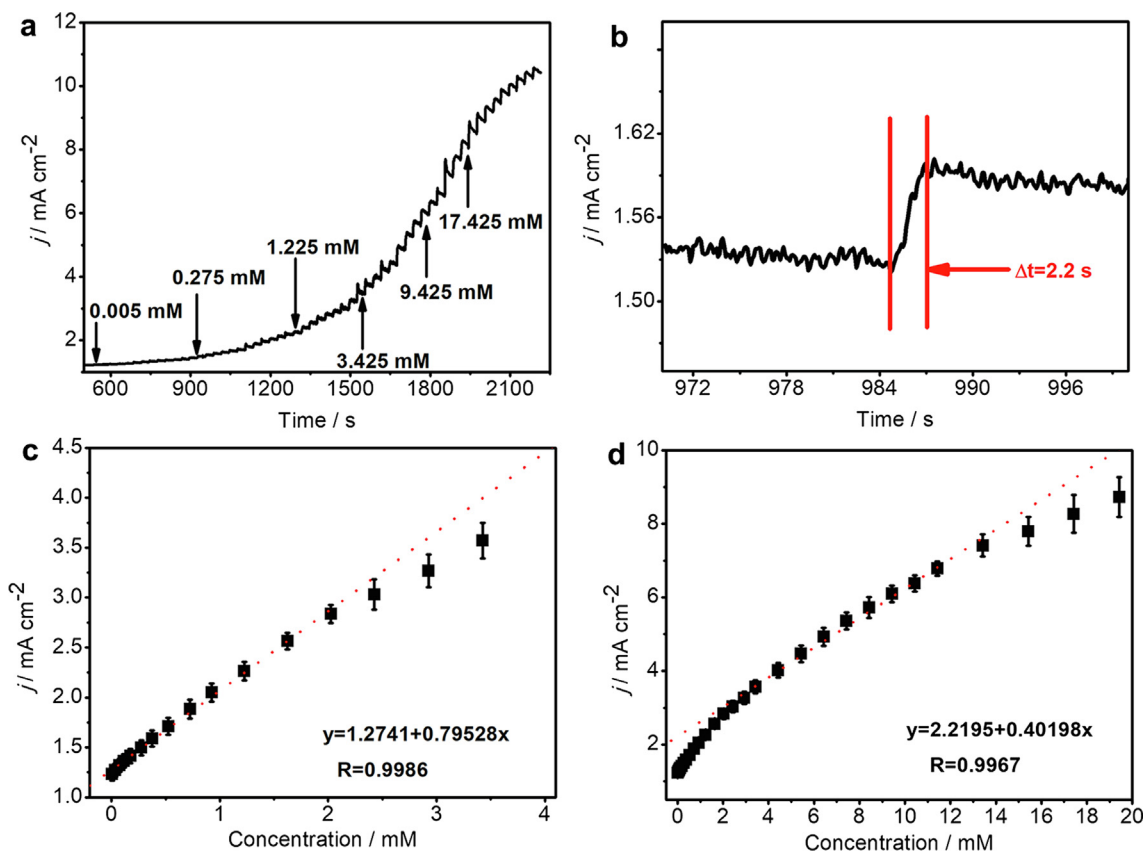


Fig. 7. (a) The amperometric j - t curve recorded with the successive addition of glucose into 0.1 M NaOH measured at +0.60 V vs. Ag/AgCl, (b) the response time toward glucose oxidation detection, (c) calibration plot in the low glucose concentration range and (d) calibration plot in the high glucose concentration range for the $\text{Co}_7\text{Fe}_3/\text{NPCSs}/\text{Nafion}/\text{GCE}$. Three replicate experiments were performed to investigate the calibration plots.

GCE, $\text{CoFe}/\text{NPCSs}/\text{Nafion}/\text{GCE}$, $\text{Co}_7\text{Fe}_3/\text{NPCSs}/\text{Nafion}/\text{GCE}$ and $\text{Co}/\text{NPCSs}/\text{Nafion}/\text{GCE}$ were determined to be 5.0 μM , 5.0 μM , 1.0 μM and 1.0 μM , respectively. The $\text{Co}_7\text{Fe}_3/\text{NPCSs}/\text{Nafion}/\text{GCE}$ has the lowest detection limit of the other electrodes. The linear low-concentration ranges are 0.005–1.40 mM, 0.005–1.55 mM, 0.001–2.20 mM and 0.001–1.80 mM for the $\text{Fe}_2\text{C}-\text{Co}_3\text{Fe}_7/\text{NPCSs}/\text{Nafion}/\text{GCE}$, $\text{CoFe}/\text{NPCSs}/\text{Nafion}/\text{GCE}$, $\text{Co}_7\text{Fe}_3/\text{NPCSs}/\text{Nafion}/\text{GCE}$ and $\text{Co}/\text{NPCSs}/\text{Nafion}/\text{GCE}$, respectively. The corresponding sensitivity values calculated for the low-concentration ranges are 272.32 $\mu\text{A mM}^{-1} \text{cm}^{-2}$, 655.10 $\mu\text{A mM}^{-1} \text{cm}^{-2}$, 795.28 $\mu\text{A mM}^{-1} \text{cm}^{-2}$, and 669.58 $\mu\text{A mM}^{-1} \text{cm}^{-2}$ for the $\text{Fe}_2\text{C}-\text{Co}_3\text{Fe}_7/\text{NPCSs}/\text{Nafion}/\text{GCE}$, $\text{CoFe}/\text{NPCSs}/\text{Nafion}/\text{GCE}$, $\text{Co}_7\text{Fe}_3/\text{NPCSs}/\text{Nafion}/\text{GCE}$, and $\text{Co}/\text{NPCSs}/\text{Nafion}/\text{GCE}$, respectively. In the high glucose concentration ranges (Figs. 7d and S12), the linear ranges and sensitivities are 1.40 to 7.00 mM and 114.73 $\mu\text{A cm}^{-2} \text{mM}^{-1}$ (for the $\text{Fe}_2\text{C}-\text{Co}_3\text{Fe}_7/\text{NPCSs}/\text{Nafion}/\text{GCE}$), 1.55 to 8.00 mM and 306.99 $\mu\text{A cm}^{-2} \text{mM}^{-1}$ (for the $\text{CoFe}/\text{NPCSs}/\text{Nafion}/\text{GCE}$), 2.20 to 14.00 mM and 401.98 $\mu\text{A cm}^{-2} \text{mM}^{-1}$ (for the $\text{Co}_7\text{Fe}_3/\text{NPCSs}/\text{Nafion}/\text{GCE}$), and 1.80 to 11.60 mM and 359.45 $\mu\text{A cm}^{-2} \text{mM}^{-1}$ (for the $\text{Co}/\text{NPCSs}/\text{Nafion}/\text{GCE}$), respectively. It is obvious that whether in the low or high glucose concentration ranges, $\text{Co}_7\text{Fe}_3/\text{NPCSs}/\text{Nafion}/\text{GCE}$ exhibits the best glucose detection parameters, including the lowest detection limit, widest linear range, and largest detection sensitivity. As shown in Table S1, compared with the other previously reported Co-based glucose sensors, including Co_3O_4 nanoparticles [37], graphene-cobalt oxide [38], $\text{CuCo}-\text{CFs}$ [2], $\text{Co}_4\text{N}-\text{NSs}$ [12], $\text{Co}_2\text{N}_{0.67}\text{NSs}$ [39], $\text{Co}_3\text{N NW}$ [40], CoP NA [41] and $\text{CoSe}-\text{rGO}$ [42], the $\text{Co}_7\text{Fe}_3/\text{NPCSs}$ catalyst prepared in this work has much wider linear concentration range for glucose detection. Meanwhile, the $\text{Co}_7\text{Fe}_3/\text{NPCSs}$ catalyst developed in this work also has a shorter response time than most of the selected Co-based glucose sensors.

3.4. Interference analysis

Due to the complex compositions of blood, a challenge in glucose detection is eliminating the influences of various interfering reagents. Fig. 8a exhibits the CV curves of the $\text{Co}_7\text{Fe}_3/\text{NPCSs}$ in a 0.1 M NaOH solution containing 1.0 mM glucose prepared with (red line) and without (black line) 0.15 M NaCl. It is obvious that the $\text{Co}_7\text{Fe}_3/\text{NPCSs}$ display an excellent anti-interference capability toward Cl^- . We further investigated the anti-interference ability of the $\text{Co}_7\text{Fe}_3/\text{NPCSs}$ toward other interfering reagents (AA, DA, LA, GA, and UA) (Fig. 8b). One can see that the $\text{Co}_7\text{Fe}_3/\text{NPCSs}$ produce negligible responses by adding 0.1 mM interfering reagents to the 0.1 M NaOH solution containing 1.0 mM glucose. The above-mentioned test results indicate the perfect anti-interference ability of the $\text{Co}_7\text{Fe}_3/\text{NPCSs}$ toward other coexisting electroactive species during glucose detection.

3.5. Reproducibility and stability

We also determined the reproducibility of the ability of $\text{Co}_7\text{Fe}_3/\text{NPCSs}$ to detect glucose oxidation by comparing the current responses of 6 different modified electrodes toward 0.15 mM glucose (as shown in Fig. 8c). Three replicate experiments were performed for each modified electrode, and the amperometric response for each electrode is the average value of the three replicates. It is clear that the amperometric responses of the 6 different $\text{Co}_7\text{Fe}_3/\text{NPCSs}/\text{Nafion}/\text{GCEs}$ toward 0.15 mM glucose reveal the superior reproducibility of the electrodes with a relative standard deviation (R.S.D.) of 3.74%. Finally, Fig. 8d exhibits the relationships between the current percentages and storage time for one of the $\text{Co}_7\text{Fe}_3/\text{NPCSs}/\text{Nafion}/\text{GCEs}$. Three replicate experiments were

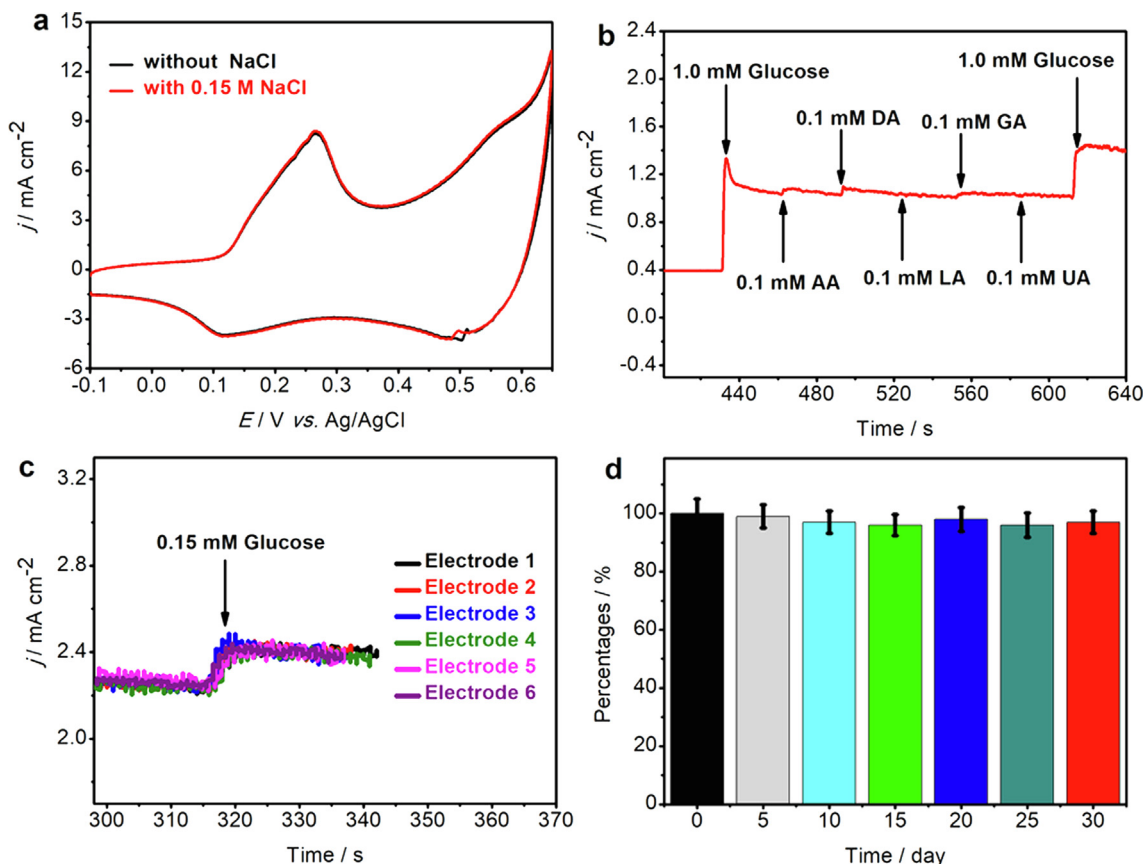


Fig. 8. (a) CV curves of the Co₇Fe₃/NPCSSs/Nafion/GCE measured in a 0.1 M NaOH solution containing 1.0 mM glucose prepared without (black line) and with (red line) 0.15 M NaCl (scan rate: 50 mV s⁻¹). (b) The amperometric j - t curve of the Co₇Fe₃/NPCSSs/Nafion/GCE measured with the successive addition of 1.0 mM glucose, 0.1 mM interference compounds (AA, DA, LA, GA, and UA), and 1.0 mM glucose at +0.6 V vs. Ag/AgCl. (c) The amperometric responses of 6 different Co₇Fe₃/NPCSSs/Nafion/GCEs toward 0.15 mM glucose at +0.6 V vs. Ag/AgCl. (d) The stability test of the same Co₇Fe₃/NPCSSs/Nafion/GCE toward glucose detection measured every five days for one month. (For interpretation of the references to colour in this figure legend, the reader is referred to the web version of this article.)

performed for each test; the current value was the average value of the three replicates. One can see that the Co₇Fe₃/NPCSSs/Nafion/GCE maintains at least 96.5% of the initial current value after one month, demonstrating an excellent long-term stability.

3.6. Real sample analysis

Finally, we explored the feasibility of detecting glucose in human serum samples by using the Co₇Fe₃/NPCSSs/Nafion/GCE. As shown in Table S2, the corresponding detection results are satisfying, and all the detection results agree with those measured by utilizing the glucose biosensor used by a local hospital (where each human serum sample was tested four times). The corresponding detection results indicate that the glucose detection ability of our Co₇Fe₃/NPCSSs/Nafion/GCE is not only appropriate for testing standard glucose samples but also suitable for testing real human serum samples.

4. Conclusion

In summary, we have successfully synthesized ultrafine Co₇Fe₃ alloy nanoparticles embedded in nitrogen-doped porous carbon nanosheets (denoted as Co₇Fe₃/NPCSSs), as reported in this paper. The structural characterization results indicated that large amounts of *meso*/macroporous channels and carbon edges/defects were uniformly dispersed on the Co₇Fe₃/NPCSSs, which led to a large BET surface area. Due to the nanoconfined effect of the mesopores and carbon edges/defects, the ultrafine Co₇Fe₃ alloy nanoparticles were uniformly dispersed along the surfaces of the Co₇Fe₃/

NPCSSs. When utilized for glucose oxidation catalysis and detection, the optimal Co₇Fe₃/NPCSSs catalyst exhibited larger glucose catalysis currents, a faster response time, a higher detection sensitivity and a wider linear range than the other control samples (Fe₂C/NPCSSs, Fe₂C-Co₃Fe₇/NPCSSs, CoFe/NPCSSs and Co/NPCSSs). Especially, the Co₇Fe₃/NPCSSs catalyst prepared in this work has much wider linear concentration range for glucose detection (0.001 to 14.00 mM) compared with the other previously reported Co-based glucose sensors, including Co₃O₄ nanoparticles (0.005–0.80 mM) [37], Co₄N-NSs (0.60 to 10.00 mM) [12], Co₂N_{0.67} NSs (0.01–8.00 mM) [39], and CoSe-rGO (up to 10.00 mM) [42]. Our optimal Co₇Fe₃/NPCSSs catalyst also displays a shorter response time and a wider linear range than the other Co-based glucose sensors reported previously. Furthermore, the Co₇Fe₃/NPCSSs also display a perfect selectivity and stability for glucose detection. The Co₇Fe₃/NPCSSs also exhibit a satisfying efficiency for glucose detection in human serum samples. The excellent glucose sensor activity of the optimal Co₇Fe₃/NPCSSs can be attributed to the synergistic effect of excellent electron conductivity, perfect porosity and abundant Co⁴⁺ active sites. Finally, our low-cost synthetic strategy can advance research in designing three-dimensional hierarchical *meso*/macroporous noble-metal-free catalysts without using any tedious steps or templates.

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Appendix A. Supplementary material

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