

First-Principles Calculations on the Adsorption Behavior of Amino Acids on a Titanium Carbide MXene

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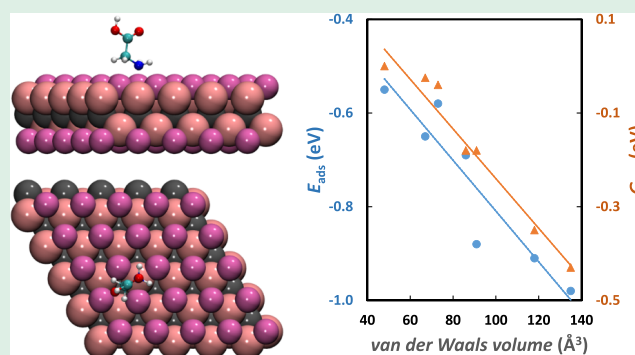
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ABSTRACT: Due to their vast range of promising biomedical and electronic applications, there is a growing interest in bioinorganic lamellar nanomaterials. MXenes are one such class of materials, which stand out by virtue of their demonstrated biocompatibility, pharmacological applicability, energy storage performance, and feasibility as single-molecule sensors. Here, we report on first-principles predictions, based on density functional theory, of the binding energies and ground-state configurations of six selected amino acids (AAs) adsorbed on O-terminated two-dimensional titanium carbide, Ti_2CO_2 . We find that most AAs (aspartic acid, cysteine, glycine, and phenylalanine) prefer to adsorb via their nitrogen atom, which forms a weak bond with a surface Ti atom, with bond lengths of around 2.35 Å. In contrast, histidine and serine tend to adsorb parallel to the MXene surface, with their α carbon about 3 Å away from it. In both adsorption configurations, the adsorption energies are on the order of the tenths of an electronvolt. In addition, we find a positive, nearly linear correlation between the binding energy of each studied AA and its van der Waals volume, which suggests an adsorption dominated by van der Waals forces. This relationship allowed us to predict the adsorption energies for all of the proteinogenic AAs on the same Ti_2CO_2 MXene. Our analysis additionally shows that in the parallel adsorption mode there is a negligible transfer of charge density from the AA to the surface but noticeable in the N-bonded adsorption mode. In the latter, the isosurfaces of charge density differences show accumulation of shared electrons in the region between N and Ti, confirming the predicted N–Ti bond. The moderate adsorption energy values calculated, as well as the preservation of the integrity of both the AAs and the surface upon adsorption, reinforce the capability of Ti_2CO_2 as a promising reusable biosensor for amino acids.

KEYWORDS: 2D materials, adsorption, amino acid, biosensors, clean MXenes, density functional theory, titanium carbide



INTRODUCTION

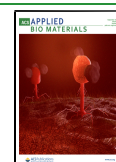
The isolation of graphene in 2004 triggered the beginning of an era of extensive study of dozens of families of two-dimensional (2D) materials.¹ In 2011, a new class of 2D materials, comprising transition-metal nitrides, carbides, or carbonitrides, was reported for the first time.² Since these materials are obtained from MAX phases, with details on their synthesis and properties given in a recent review,³ they were named MXenes. Briefly, MXenes are constituted by atomic layers of early-transition-metal elements (M) intercalated by atomic layers of carbon and/or nitrogen (X), with the general formula $\text{M}_{n+1}\text{X}_n\text{T}_x$ ($n = 1-3$), where T_x represents surface terminations (usually O, OH, F, and H moieties) attached to the external M layers.⁴ Among other properties, MXenes are characterized by high conductivity, high oxidation resistance, hydrophilicity, and easily tunable surface terminations.⁴⁻⁶ These properties make MXenes appropriate for energy storage applications, such as batteries,⁷ supercapacitors,⁶ and sensors;⁸ they have recently been proposed for carbon capture

technologies⁹⁻¹¹ and are employable as heterogeneous catalysts.¹²⁻²⁰ The present work focuses on the possible biosensing capabilities of these materials taking the O-terminated titanium carbide Ti_2C , hereafter denoted as Ti_2CO_2 , as a prototype MXene. Although MXenes without surface terminations have been synthesized,²¹ oxygen termination seems to be one of the most common ones, in special under oxidative environments. Moreover, unlike its hydroxyl counterpart, the oxygen termination allows the effective development of a 2D material, as it does not lead to the formation of hydrogen bonds coupling MXene layers to each other. Interestingly, Ti_2CO_2 has been shown to be suitable for

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detecting amino acids (AAs) (namely, arginine) *via* adsorption,²² and its interaction with lysozyme has been experimentally studied.²³ DNA molecules have also been shown to be able to cross Ti_2CO_2 nanopores, and DNA transport studies through MXene membranes suggest the potential of this class of 2D materials for DNA translocation and single-molecule sensing applications.²⁴ Moreover, the fact that Ti_2CO_2 contains the most biocompatible transition metal, makes it readily versatile for biomedical applications.²⁵

Despite the demonstrated potential of Ti_2CO_2 as a biosensor,²⁶ to the best of our knowledge, no proofs of concept, aside from the ones mentioned, have been performed, and very little theoretical work has been done regarding the adsorption of amino acids on MXenes, in general, and on this MXene, in particular. A convenient theoretical framework to address this subject is density functional theory (DFT). In fact, DFT-based calculations have been used to determine the binding energy of glycine to $\text{Ti}_3\text{C}_2\text{O}_2$ ²⁷ and the adsorption behavior of valine, arginine, and aspartic acid on other 2D materials such as graphene and hexagonal boron nitride (BN).²⁸ In the latter, it was further shown that dispersion corrections to the energy are essential for an accurate description of the adsorption process.

In the present work, we focus on the adsorption of six amino acids (AAs) on Ti_2CO_2 as described by means of DFT-based calculations. The chosen amino acids are aspartic acid (ASP), cysteine (CYS), glycine (GLY), histidine (HIS), phenylalanine (PHE), and serine (SER). Alanine (ALA) was also studied for validation purposes (see below). These were chosen to cover the whole range of polarity/acidity values of amino acids. In particular, we chose amino acids with an aliphatic (GLY and ALA), one aromatic (PHE), an acidic (ASP), a basic (HIS), a hydroxylic (SER), and one sulfur-containing (CYS) side chain. In the rest of this paper, we first describe the computational details of our calculations and then the respective results and finally provide the study conclusions.

■ COMPUTATIONAL METHODS

The DFT-based calculations were carried out using the Vienna *ab initio* simulation package (VASP).^{29–32} The exchange–correlation functional used was the one prescribed by Perdew, Burke, and Ernzerhof (PBE) within the generalized gradient approximation (GGA).³³ with dispersion corrections included *via* the so-called D3 method with Becke–Johnson damping,^{34,35} since dispersion is vital for the correct quantitative sizing of AA adsorption on surfaces.^{36–38} Dipole corrections in the direction perpendicular to the slab were also taken into account.³⁹ The explicit treatment of core electrons in the Kohn–Sham equations was avoided by employing the projector-augmented wave (PAW) method,⁴⁰ which allowed including their effect on the valence electron density. The Kohn–Sham equations were solved using a plane-wave (PW) basis set, with an energy cutoff of 600 eV, until the total energy difference between two consecutive iterations was less than 10^{-7} eV. Structural relaxations stopped when the total force acting on each atom became smaller than $0.005 \text{ eV}\cdot\text{\AA}^{-1}$. The Brillouin zone was sampled using a Γ -centered grid with a distribution of $2 \times 2 \times 1$ k -points.⁴¹ The choice of these values was based on a series of convergence tests, ensuring that relative energies were converged within less than $1 \text{ meV}\cdot\text{atom}^{-1}$. Spin polarization was not considered, as preliminary tests showed that it would leave the results unaltered.

We used rhombic supercells of the same shape and size for all of the adsorbed AAs on the Ti_2CO_2 MXene (Figure 1). The bulk supercell contained $p(5 \times 5)$ unit cells of Ti_2CO_2 , with D_{3d} symmetry. The calculated lattice parameter and slab thickness, $a = 3.015 \text{ \AA}$ and $d = 4.46 \text{ \AA}$, respectively, are in good agreement with the values $a = 3.04$

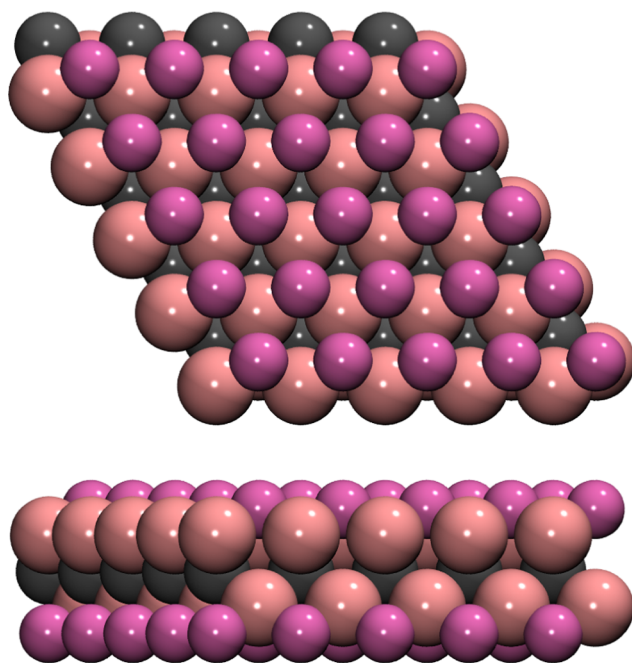


Figure 1. Top and side views of the Ti_2CO_2 supercell. The spheres represent atoms, and different colors represent different chemical elements: C is shown in gray, O is shown in magenta, and Ti is shown in pink.

$\text{\AA}$ and $d = 4.45 \text{ \AA}$ obtained previously at the PBE level.⁴² To avoid interactions in the direction perpendicular to the plane of the slab due to the periodic boundary conditions, the cell height was fixed at $24 \text{ \AA}$, accounting for approximately $19.5 \text{ \AA}$ of vacuum, to be partially filled by the AAs.

Only L-AAs were sampled, given the achiral character of the Ti_2CO_2 surfaces. Several orientations of the molecules relative to the Ti_2CO_2 surface were studied as detailed in Figure S1. To speed up the process of finding the most stable orientations on the MXene surface, the semiempirical extended tight-binding program package (xtb, version 6.2.3)^{43,44} was used for an initial screening. In these calculations, using periodic boundary conditions and default parameters, the Ti, C, and O atoms of the MXene were kept frozen while the atomic positions of the AAs were fully optimized. In general, the three most stable structures thus found were then used as initial starting points in the subsequent full-optimization runs with the VASP code. The ground-state geometry of each amino acid adsorbed on the surface was then established by comparing the total energy of the local minimum configurations of each AA or, equivalently, by picking the one with the most negative adsorption energy, E_{ads} , defined as

$$E_{\text{ads}} = E_{\text{AA/MXene}} - (E_{\text{MXene}} + E_{\text{AA}}) \quad (1)$$

where $E_{\text{AA/MXene}}$ is the total energy of the Ti_2CO_2 supercell containing the adsorbed AA, E_{AA} is the energy of the isolated AA, calculated in the same unit cell, and E_{MXene} is the total energy of the pristine Ti_2CO_2 slab. As defined, negative values of E_{ads} refer to stable adsorptions.

The local-minimum character of the ground-state configurations was further verified by the normal modes of vibrational analysis of the adsorbed AAs, ensuring that all of the vibrational frequencies were real. This was done by diagonalization of the corresponding block of the Hessian matrix with elements obtained by finite differences of analytical gradients with displacements of $0.015 \text{ \AA}$. We investigated the effect of including the degrees of freedom of the surface atoms closest to the adsorbed AAs in frequency calculations and found that the vibrational modes of the surface are unchanged in the presence of the adsorbed AAs. As we will see, this is to be expected, given the weak adsorbate–adsorbent interaction. For this reason, the surface contribution was neglected in subsequent frequency calculations. The

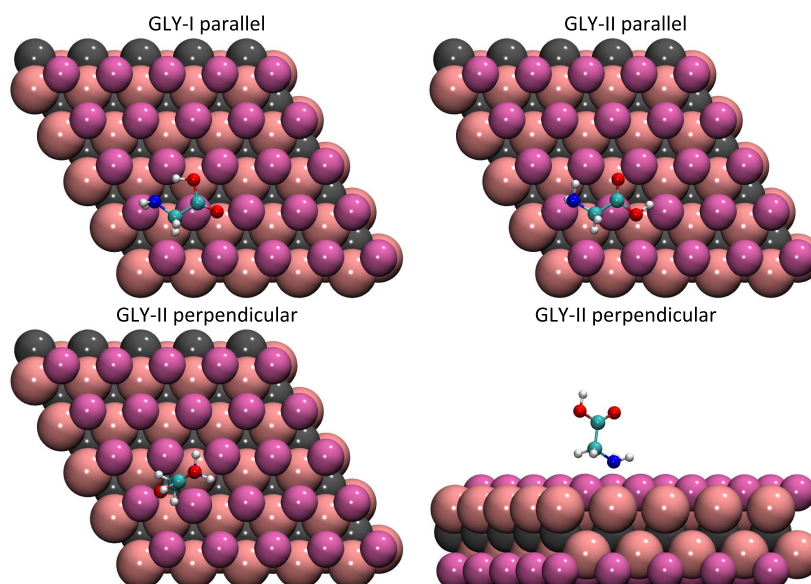


Figure 2. Several stable adsorbed configurations of GLY on the Ti_2CO_2 MXene surface. The top panels depict parallel adsorption of GLY in its two most stable conformations (I and II).⁵³ The bottom panels show the top (left) and side (right) views of the perpendicular adsorption configuration of GLY-II. The color code for the MXene atoms is the same as in Figure 1. The color code for the smaller spheres (Gly atoms) is as follows: H is shown in white, C is shown in cyan, N is shown in blue, and O is shown in red.

AA frequencies were then used to calculate the partition function, and transition-state theory was employed to evaluate the Gibbs free energy for the adsorption process. Representing schematically the process as



where * stands for a surface active site, the Gibbs free energy of adsorption is given by

$$G_{\text{ads}} = E_{\text{ads}} - k_{\text{B}}T \ln \left(\frac{q_{\text{AA}^*}}{q_{\text{AA}_{(\text{g})}}} \right) \quad (3)$$

where k_{B} is the Boltzmann constant, T is the absolute temperature, and q is the partition function of the molecule in the gas phase, $\text{AA}_{(\text{g})}$, and in its adsorbed form, AA^* .⁴⁵ The partition function of the molecule in the gaseous phase⁴⁶ is the product of the vibrational, rotational, translational, and electronic contributions ($q = q_{\text{vib}} \cdot q_{\text{rot}} \cdot q_{\text{trans}} \cdot q_{\text{el}}$), while in its adsorbed phase, only the vibrational contribution is included, given that the molecular translations and rotations become frustrated because of the adsorption bond. The vibrational partition function can be calculated by knowing the frequencies ν_i of the normal vibrational modes (Tables S1 and S2 of the Supporting Information (SI))

$$q_{\text{vib}} = \prod_i \frac{\exp(-h\nu_i/(2k_{\text{B}}T))}{1 - \exp(-h\nu_i/(k_{\text{B}}T))} \quad (4)$$

where h is the Planck constant. The rotational and translational contributions^{45,46} to the partition function are given by

$$q_{\text{rot}} \approx \frac{\sqrt{\pi}}{\sigma} \sqrt{\frac{T^3}{\Theta_{r,x} \Theta_{r,y} \Theta_{r,z}}} \quad (5)$$

and

$$q_{\text{trans}} = \left(\frac{2\pi mk_{\text{B}}T}{h^2} \right)^{3/2} \frac{k_{\text{B}}T}{p} \quad (6)$$

respectively. Here, $\sigma = 1$ is the rotational symmetry number of the AAs,⁴⁷ the $\Theta_{r,i}$ terms are the characteristic rotational temperatures around the i axes,⁴⁸ m is the mass of the molecule, and p is the AA partial pressure. In this work and unless otherwise noted, the

temperature and pressure were fixed at 298.15 K and 1 bar, respectively, and the rotational temperatures of all of these gaseous AAs were calculated from the diagonalization of their inertia tensors. The procedure for obtaining the molecular partition functions is the same as the one employed in recent studies.^{49–51} The electronic contribution to the partition function, q_{el} , can be approximated to one, given the singlet ground-state character of AAs and large energy gap to their excited states.

Finally, we investigated the possibility of electron transfer between the MXene surface and the adsorbed AAs. To this end, we calculated the Bader charges⁵² and the charge density differences between the surface with the adsorbed AA and the infinitely separated MXene and AA with the same geometry. These concepts and their interpretation are explained in detail in the next section.

RESULTS

This section is divided in two parts. We first describe in detail our results concerning the mechanism of adsorption of GLY on the titanium carbide MXene and then present the results for the remaining AAs considered, *viz.*, ALA, ASP, CYS, HIS, PHE, and SER.

Adsorption of Glycine. We studied the GLY molecule (with chemical formula $\text{NH}_2\text{CH}_2\text{COOH}$) in its two most stable conformations,⁵³ I and II, shown in the two top panels of Figure 2. These two conformations differ in the nature of the internal hydrogen bonds, *i.e.*, O–H...N in Gly-I and N–H...O in Gly-II. According to our calculations, the two molecular conformations are nearly degenerate in the gas phase, GLY-I being slightly more stable, by 51 meV, a value which is of the order of the error bar of the calculations. Conformation I was therefore taken as reference for adsorption energy calculations. Several adsorption configurations were compared, as shown in Figure S1 of the Supporting Information (SI). Here, we focus on the configurations displaying the strongest adsorption strength, shown in Figure 2.

When adsorbing parallel to the surface, GLY sits with its α carbon atom ~ 3 Å away from the plane defined by the surface O atoms. The adsorption occurs with the amine group of GLY

near a surface O atom, with one of its H atoms pointing toward that O atom. One of the H atoms in the α carbon is oriented toward the surface, while the other is pointing in the direction of the vacuum region. The carboxylic group (COOH) remains parallel to the surface. The adsorption energies for conformations I and II of GLY, both in the mode parallel to the surface, are -0.52 and -0.42 eV, respectively.

The GLY-II conformation can also adsorb on the Ti_2CO_2 surface by its nitrogen atom, keeping the plane defined by the α carbon, the amino, and the carboxyl groups, $\text{H}_2\text{N}-\text{C}_\alpha-\text{C}(=\text{O})\text{OH}$, normal to the surface (Figure 2, bottom panels). This configuration was found to be the most stable for GLY interacting with this MXene, with an adsorption energy of -0.55 eV. Note that the most stable GLY adsorption configuration on the Ti_2CO_2 surface is the same as the one reported previously on $\text{Ti}_3\text{C}_2\text{O}_2$, with the GLY adsorbate perpendicular to the surface plane and oriented with its amine moiety toward the surface.²⁷ In the case of the most favorable adsorption configuration, the nearest surface Ti atom is pulled ~ 0.35 Å out of the Ti plane and toward the amino acid, so that the N–Ti distance becomes 2.35 Å, which is the same value as that found in ref 27. The short distance and the quite large adsorption energy suggest a chemical bond between GLY and the MXene surface, as will be analyzed in detail below. The parallel adsorption of GLY shows positive values of G_{ads} around 70 meV at ambient temperature and pressure (298.15 K and 1 atm), while the perpendicular adsorption has a G_{ads} of zero.

For the sake of comparison with the results of ref 27, we briefly compare the adsorption of GLY on the $\text{Ti}_3\text{C}_2\text{O}_2$ MXene surface. We found that, as on Ti_2CO_2 , the strongest adsorption is found when the molecule is perpendicular to the surface and bound by its N atom to a Ti atom of the surface (see Figure S2), with an adsorption energy of -0.50 eV using the PBE-D3 functional or -0.57 eV with the PBE-optB86b functional. The difference between these values is within the error bar of the calculations, and therefore, they can be considered to be identical to each other and to the calculated adsorption energy on Ti_2CO_2 , suggesting that the thickness of the MXene surface has no significant effect on the mechanism or strength of AA adsorption, as also seen in other studies, e.g., on the CO_2 capture with MXenes with different widths.⁵⁴

The calculated vibrational frequencies of GLY in the gaseous phase or adsorbed on the Ti_2CO_2 are shown in Table S1. The frequencies of gaseous GLY can be compared to those in ref 55, where infrared (IR) spectra were recorded for several GLY conformers, including GLY-I, our gas-phase reference. For the gaseous phase GLY-I, our calculated frequencies are systematically lower than the measured ones, although only by 16 cm^{-1} on an average. In addition, we find two frequencies, associated with CCH bending modes, in the 1300 cm^{-1} region, which are absent from the list of measured frequencies, an effect observed in ref 55 as well. The mode related to CN stretching at 1050 cm^{-1} calculated by us is also absent from the experimental spectrum and has correspondence with the 1059 cm^{-1} frequency calculated in ref 55.

In contrast to the above results for the gas-phase GLY conformers, the calculated normal mode frequencies of the adsorbed GLY-II differ from those obtained experimentally by up to 70 cm^{-1} . The calculated frequencies (and respective IR intensities) for adsorbed GLY can be compared with the ones in the IR spectrum obtained experimentally for GLY on $\text{Ti}_3\text{C}_2\text{T}_x$.²⁷ Table S3 summarizes these comparisons, and

Figure S4 shows the IR spectrum of GLY-II adsorbed perpendicularly on Ti_2CO_2 , calculated using the method reported in ref 56. This plot shows that the modes predicted to have the highest IR intensities are the ones at 1755 , 1361 , 1276 , 1136 , 1091 , and 1028 cm^{-1} . The experimental band observed at 1640 cm^{-1} for GLY on $\text{Ti}_3\text{C}_2\text{T}_x$, which was assigned to NH_2 bending is blue-shifted by around 20 cm^{-1} upon adsorption. The calculated value is 1599 cm^{-1} , but its intensity is very low and, therefore, it is absent from the calculated spectrum. There are plenty of explanations for this discrepancy, related to the cleanness of the surface, the presence of a solvent, and the adsorption mechanism. First, here we assume that the Ti_2C surface is terminated by a single layer of O, while the experiments in ref 27 were conducted using $\text{Ti}_3\text{C}_2\text{T}_x$, where the T_x termination is mostly unknown and may include $-\text{O}$, $-\text{OH}$, or $-\text{F}$. Second, the calculations are performed assuming ultrahigh vacuum, which differs significantly from the experimental conditions in ref 27. Third, the most favored adsorption configuration of GLY might not be as simple as a single adsorbed molecule, isolated from all of the other GLY molecules.

Adsorption of Other Amino Acids. After the GLY test case, we studied the adsorption mechanism of ASP, CYS, HIS, PHE, and SER on Ti_2CO_2 . The most stable adsorption configuration of each AA was found after energetically comparing orientations analogous to the ones analyzed for GLY, as shown in Figures 2 and S1. As in the case of GLY, the two most stable adsorption configurations of each AA are the parallel one and the one that allows formation of a chemical bond between the nitrogen atom of the AA and the nearest surface Ti atom. Figure 3 shows the optimized geometries of each adsorbed AA, both parallel and N-bonded to the surface.

The parallel adsorptions occur with the nearest atom (a hydrogen in all of the cases) about 2 Å away from substrate, which remains virtually unaffected by the presence of the AA. The amino groups of ASP and SER lie near an O surface atom with one of its H atoms pointing toward that O atom, while the α C stays near another surface O atom. This contrasts with CYS and HIS where the amine group is on top of a Ti atom and the α C lies above a carbon hollow site of the surface. In the case of PHE, the N and the α C are both near Ti atoms of the surface and the H of the α C is the atom closest to the surface. Interestingly, the H atom in the α carbon of CYS, HIS, and PHE points toward the surface, while in ASP and SER, the same atom points away from the surface. However, the energy differences between the configurations with the H atom in the α carbon pointing toward or away the surface are below 0.1 eV. Regarding the N-bonded adsorption, from inspection of the right-hand-side panels of Figure 3, one readily notices that, like in the case of GLY adsorption, the nearest Ti atom is drawn toward the nitrogen atom of all of the AAs. Nonetheless, unlike GLY, the larger AAs do not remain truly perpendicular to the surface but instead attempt to return to their parallel orientation, while keeping the N and Ti atoms close to each other. The possibility of an N–Ti bond is analyzed in detail below. As we will see, this configuration is advantageous in some AAs, while others prefer the parallel alignment. Table 1 shows relevant calculated properties of the adsorbed AAs. All of the adsorption energies (E_{ads}) are negative, regardless of the relative orientation of the AA and the surface, indicating favorable adsorption in both situations. Interestingly, some AAs prefer parallel adsorption, while others prefer the N-

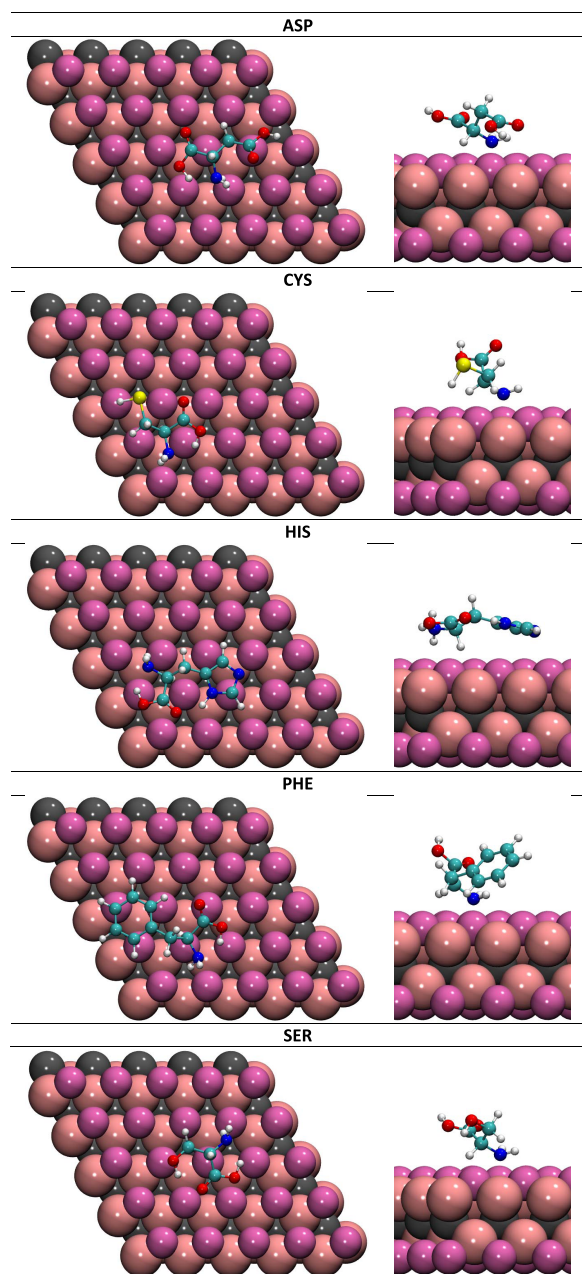


Figure 3. Ground-state adsorbed configurations of ASP, CYS, HIS, PHE, and SER. A top view of the parallel adsorption is shown on the left, and a side view of the N-bonded adsorption is shown on the right. The color code is the same as that of Figure 2, and S is shown in yellow.

bonded one; furthermore, on some AAs, the relative stability is reversed at ambient temperature and pressure.

For parallel adsorption, the distances between the α carbon of each AA and the carbon layer of Ti_2CO_2 ($d_{\alpha-\text{O}}$) were calculated in the direction perpendicular to the surface. There is clearly no correlation between this distance and the adsorption energies, as the AAs with the most similar adsorption energies, ASP and CYS, are the ones with the highest and lowest $d_{\alpha-\text{O}}$, respectively. The same lack of correlation is observed between the adsorption energies and N–Ti distances ($d_{\text{N-Ti}}$) in the case of N-bonded adsorption. However, the overall results also hint that the largest AAs adsorb more strongly to the surface. As can be seen in Figure

S3, the parallel adsorption stability of the AAs increases in the order $\text{GLY} < \text{SER} < \text{CYS} < \text{ASP} < \text{HIS} < \text{PHE}$, which coincides with the order of molar masses of the AAs. This is to be expected for an interaction dominated by van der Waals (vdW) forces, which is favored by a large number of contacts with subtle differences found in the calculated infrared spectra (Figure S6) that seem to originate from the composition of the AA lateral chain. To verify this assertion, we investigated the correlation between the calculated binding energies of the AAs and their van der Waals volumes (V).⁵⁷ The linear correlation is evident from Figure 4, via the represented linear regression line and the respective correlation coefficient of $R^2 = 0.87$. The Gibbs free energies of adsorption (ΔG_{ads}) also display a good linear correlation with the van der Waals volumes, with $R^2 = 0.96$. This correlation reinforces the claim regarding the dominant role of van der Waals interactions in the adsorption process, as the strength of these interactions increases with the size of the interacting molecules. The adsorption energy of a seventh amino acid (alanine—ALA) was also calculated and included in the plot, with its most stable (N-bonded) adsorption configuration shown in Figure 5. We found that alanine fits the trend found for the original six AAs (Figure 4) with a difference between the calculated E_{ads} value (−0.65 eV, Table 1) and the one predicted with the fitted equation (−0.64 eV, Table 2) of only 0.01 eV. The lines represented for E_{ads} and G_{ads} have very similar slopes, suggesting that the extra entropy-related term involved in the calculation of G_{ads} is practically independent of the AA and approximately equal to 0.56 eV. Using the calculated relationships $E_{\text{ads}} = f(V_{\text{vdW}})$ and $G_{\text{ads}} = f(V_{\text{vdW}})$ in Figure 4, we predicted the adsorption energies and Gibbs free energies of adsorption of the remaining proteinogenic amino acids on Ti_2CO_2 , as shown in Table 2. The adsorption energies are between −0.53 and −1.15 eV, with the largest AA, tryptophan, showing the strongest adsorption.

The adsorptions of GLY and PHE on the reduced Ti_2C surface, i.e., that obtained from the Ti_2CO_2 MXene upon removal of the terminating oxygen atoms, were also analyzed in this work. As expected, the high reactivity of the bare surface leads to much stronger adsorptions (with E_{ads} of −3.04 and −8.05 eV, respectively) and to significantly distorted molecular structures (Figure S5). This suggests activation of the molecule that would be sensitive to suffer bond dissociation.

Let us now analyze the possibility of electronic density transfer between the surface and each of the adsorbed molecules. The outcome of Bader charge analysis⁵² shows an increase of the charge density on the surface, regardless of which AA was adsorbed and of its orientation relative to the surface. However, in the parallel adsorption, the magnitude of this change is small, amounting to less than 0.1 e , and in some cases could even be assigned to uncertainties of numerical integration in the Bader atomic basins. In contrast, the surface Bader charge differences are expectedly higher when the adsorption occurs via N–Ti bonding and are no longer negligible. In fact, the N-bonded adsorption charge variations are around an order of magnitude higher than the ones related to parallel adsorption, supporting the conclusion that a chemical interaction takes place between N and Ti. To test this hypothesis, we calculated the N–Ti bond order for each adsorbed AA (see Table 1) and found values around 0.15, representative of weak bonding and displaying very good correlation ($R^2 = 0.9975$) with the N–Ti distances. No correlation is found between the adsorption energies and the

Table 1. Relevant Physicochemical Quantities for the Adsorption of the Studied AAs on Ti_2CO_2 ^a

AA	V_{vdW} ($\text{\AA}^3$) ^b	parallel adsorption				N-bonded adsorption				
		$d_{\alpha-\text{O}}$ ($\text{\AA}$) ^c	E_{ads} (eV)	G_{ads} (eV)	ΔQ (e)	$d_{\text{N-Ti}}$ ($\text{\AA}$) ^d	N-Ti bond ^e	E_{ads} (eV)	G_{ads} (eV)	ΔQ (e)
ASP	91	3.42	−0.79	−0.23	0.00	2.33	0.169	−0.88	−0.18	0.19
CYS	86	2.79	−0.67	−0.18	0.09	2.42	0.137	−0.69	−0.04	0.18
GLY	48	3.08	−0.53	0.00	0.01	2.35	0.161	−0.55	0.07	0.19
HIS	118	2.85	−0.91	−0.35	0.16	2.33	0.168	−0.76	−0.14	0.23
PHE	135	3.00	−0.94	−0.43	0.07	2.35	0.161	−0.98	−0.34	0.21
SER	73	3.46	−0.60	−0.04	0.01	2.33	0.169	−0.47	0.14	0.20
ALA	67	3.53	−0.55	−0.02	0.01	2.33	0.170	−0.65	−0.03	0.21

^aThe data include the adsorption energy (E_{ads}), Gibbs free energy of adsorption (G_{ads}) of the AA at 298 K and 1 bar, and the Bader charge (ΔQ) of the surface unit cell in the presence of each adsorbed amino acid. For each AA, the lowest E_{ads} and the lowest G_{ads} among the parallel and N-bonded adsorptions are highlighted in bold. ^bThe van der Waals volumes (V) of the AAs were taken from ref 57. ^cDistance between the α carbon of the AA and the nearest oxygen surface layer. ^dDistance between the N atom of the AA and its nearest surface Ti atom ($d_{\text{N-Ti}}$). ^eBond order between the N atom of the AA and its nearest surface Ti atom, calculated using the method reported in ref 58.

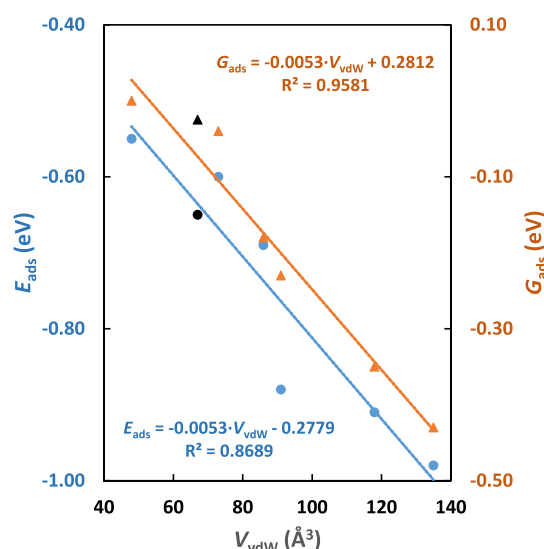


Figure 4. Plot of the calculated adsorption energies (blue dots) and adsorption Gibbs energies (orange triangles) of the six studied amino acids as a function of their van der Waals volumes and their respective linear regression lines and fitted equations. The black dot/triangle corresponds to the adsorption energy/Gibbs energy of a seventh amino acid, alanine, not included in the fitting (see the text for details).

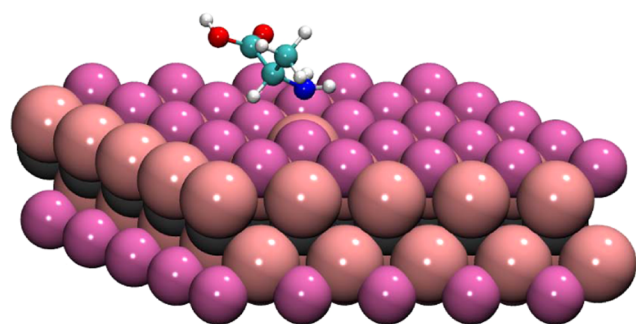


Figure 5. Ground-state configuration of ALA adsorbed on the Ti_2CO_2 surface. Color code is the same as in Figure 2.

charge transferred from the adsorbate to the surface. The calculated projected density of states on the surface and AA bands are provided in Figures S7 and S8 for parallel and N-bonded adsorptions, respectively. In the case of the bare surface, the calculated band gap of 0.214 eV compares well

Table 2. Calculated Adsorption Energies (E_{ads}) and Gibbs Free Energies of Adsorption (G_{ads}) of the twenty Amino Acids on Ti_2CO_2 , Extrapolated from the Relationship $E_{\text{ads}} = -0.0053 \cdot V_{\text{vdW}} - 0.2779$ and $G_{\text{ads}} = -0.0053 \cdot V_{\text{vdW}} + 0.2812$

AA	V_{vdW} ($\text{\AA}^3$)	E_{ads} (eV)	G_{ads} (eV)
ALA	67	−0.64	−0.07
ARG	148	−1.07	−0.50
ASN	96	−0.79	−0.23
ASP	91	−0.76	−0.20
CYS	86	−0.74	−0.17
GLN	114	−0.89	−0.32
GLU	109	−0.86	−0.30
GLY	48	−0.53	0.03
HIS	118	−0.91	−0.34
ILE	124	−0.94	−0.38
LEU	124	−0.94	−0.38
LYS	135	−1.00	−0.43
MET	124	−0.94	−0.38
PHE	135	−1.00	−0.43
PRO	90	−0.76	−0.20
SER	73	−0.66	−0.11
THR	93	−0.77	−0.21
TRP	163	−1.15	−0.58
TYR	141	−1.03	−0.47
VAL	105	−0.84	−0.27

with the literature value,⁵⁹ 0.24 eV, that was obtained with the PBE functional. Comparing the two adsorption modes, it is seen that many AA bands in the N-adsorbed cases are in general shifted to more negative energy values than for the parallel adsorption, which is a consequence of the more favorable adsorption in the former.

Finally, we built three-dimensional (3D) representations of the charge density differences. These provide a qualitative measure of the change in the charge density at each point of real space in the region where the adsorption takes place, relative to the surface and AA isolated from each other. The result is shown in Figure 6 and reveals that some features are shared by all of the AAs. The hydrogen atom of the amine group that is directed to the surface has lower density than in the molecule in the gas form (denoted by the cyan color), in both adsorption configurations. In turn, the charge density near the oxygen atoms on the surface closest to this hydrogen atom increased (represented in gold color). On parallel adsorption, we see the opposite effect in the carboxylic

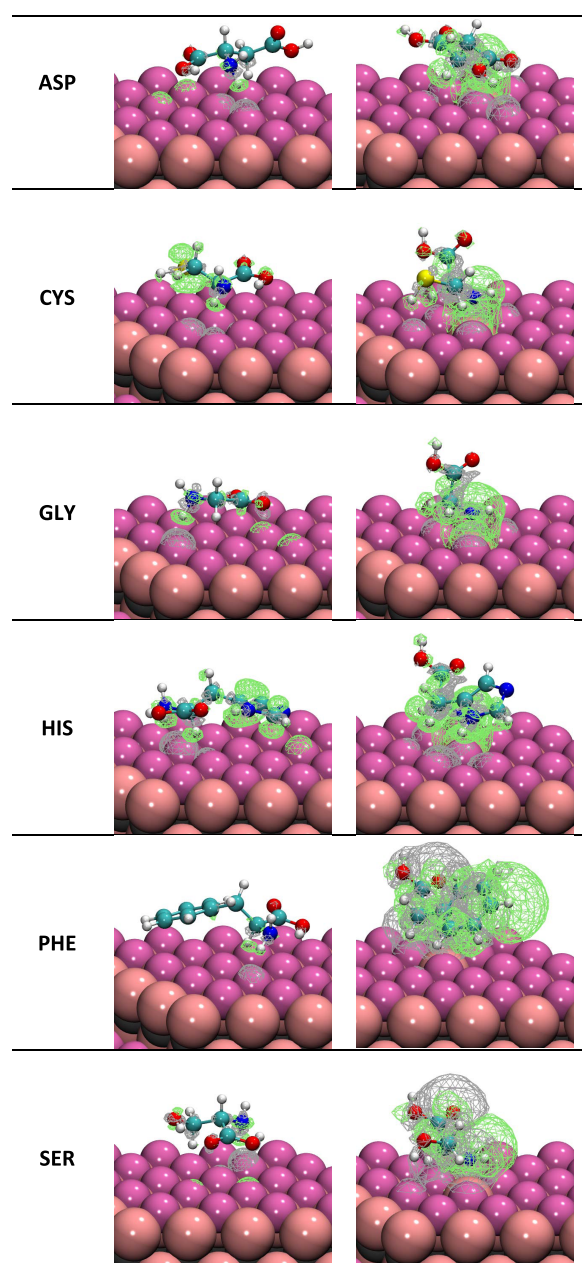


Figure 6. Isosurfaces ($0.005 \text{ e} \cdot \text{\AA}^{-3}$) of charge density differences for the AAs@MXene systems compared to the isolated surface and amino acids, in the parallel (left) and N-bonded (right) adsorption configurations. The color code for spheres is as in Figure 3. The gray (green) wired region represents an increase (decrease) in the charge density.

group of the AAs: the charge density of the oxygen atoms of this group increases, while the nearest surface oxygen atoms lose some charge density. On N-bonded adsorption, the accumulation of shared electrons between N and Ti, observed as large blobs around the N–Ti region, again indicates that a chemical bond is formed. Combining the information provided by the charge density profiles with the Bader charge calculations, one can infer that, upon parallel adsorption, the differences of charge density are the result of subtle electronic rearrangement on the surface and on the AA, which can be interpreted as a polarization of the MXene surface by the AA. The general features of our findings are thus in agreement with the ones found previously²⁸ for the adsorption of AA on

graphene and hexagonal boron nitride. In particular, both studies concluded that when the AAs orient themselves parallel to the surface (i) the surfaces present little to no deformation, (ii) there is no charge transfer between the surface of the adsorbents and the AAs, and (iii) the adsorption is stronger for larger amino acids. Interestingly, for parallel adsorption, although the atom nearest to the surface is about 1 Å closer in our case than in the case of graphene or BN, the adsorption energies on BN are an order of magnitude higher than those on Ti_2CO_2 , reflecting a higher affinity of BN toward amino acids. In fact, on Ti_2CO_2 , the atoms nearest to the surface are hydrogen atoms rather than the hetero atoms, justifying the difference in the distances mentioned.

CONCLUSIONS

Using first-principles calculations based on the DFT framework, we have predicted the adsorption behavior of amino acids on the MXene Ti_2CO_2 . At $T = 0 \text{ K}$, we found that most of the studied molecules prefer to adsorb *via* their amine N atom, chemically bound to a surface Ti atom, the latter being pulled toward the AA, sitting at a Ti–N distance of about 2.35 Å. The remaining molecules adsorb parallel to the surface, with the nearest atom and the α carbon lying around 2 and 3 Å away from the plane of oxygen atoms on the surface, respectively, while the adsorbent suffers practically no structural deformation. In both cases, the side chains of the AAs attempt to align with the plane of the surface. The absolute values of the calculated adsorption energies are between 0.5 and 1 eV to show a linear correlation with the van der Waals volumes of the AAs. Extrapolating this correlation, one can infer that no amino acid should have an adsorption energy weaker than -0.5 eV , and the AA that adsorbs most strongly on Ti_2CO_2 should be the one with the highest vdW volume, tryptophan, with an adsorption energy around -1.1 eV . At ambient temperature and pressure, the preference for parallel or N-bonded adsorption is reversed on most AAs, so that only ASP prefers to adsorb *via* N-bonding, albeit less than 0.1 eV more favorably than the parallel adsorption. This is also a clear indication of the very weak character of the Ti–N bond observed in this adsorption mode.

No significant charge transfer was found between the amino acids adsorbed with their planes parallel to the surface. However, the presence of the molecule does polarize the MXene in the nearby region: the charge density of the surface increases near the amino group and decreases near the carboxyl group of each amino acid. In contrast, N-bonded adsorption leads to a small but noticeable electron transfer from the AA to the surface, of at most 0.2 e , and to an accumulation of shared electrons between N and Ti, indicating a chemical interaction between Ti and N. The moderate adsorption energy values found, along with the lack of deformation of the amino acids and the surface, emphasize the capability of Ti_2CO_2 as an easily reusable biosensor for these molecules.

ASSOCIATED CONTENT

Supporting Information

The Supporting Information is available free of charge at <https://pubs.acs.org/doi/10.1021/acsabm.0c00621>.

Figure S1 with initial orientations of the amino acids on the Ti_2CO_2 surface; Figures S2 and S5 with the adsorbed configurations of Gly on Ti_2CO_2 and of Gly

and Phe on Ti_2C , respectively; Figure S3 with details of the phase diagrams for the amino acids adsorbed parallel to the Ti_2CO_2 surface as a function of AA partial pressures and temperature; Figures S4 and S6 with calculated infrared spectra of selected configurations of amino acids on the Ti_2CO_2 surface; Figures S7 and S8 with calculated projected density of states of selected configurations of amino acids on the Ti_2CO_2 surface; Tables S1–S3 with calculated wavenumbers for the amino acids adsorbed on the Ti_2CO_2 surface (PDF)

Zip file with optimized coordinates (ZIP)

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Notes

The authors declare no competing financial interest.

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