

Ultrasensitive and Selective Field-Effect Transistor-Based Biosensor Created by Rings of MoS₂ Nanopores

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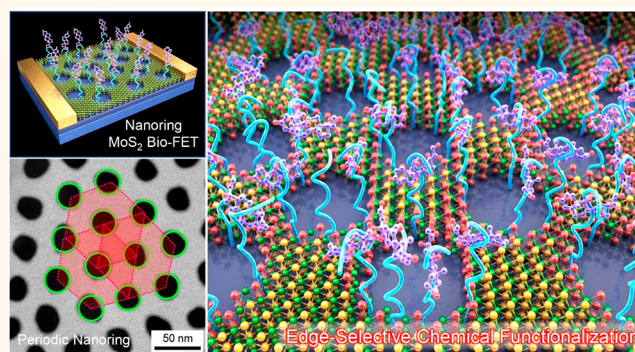
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ABSTRACT: The ubiquitous field-effect transistor (FET) is widely used in modern digital integrated circuits, computers, communications, sensors, and other applications. However, reliable biological FET (bio-FET) is not available in real life due to the rigorous requirement for highly sensitive and selective bio-FET fabrication, which remains a challenging task. Here, we report an ultrasensitive and selective bio-FET created by the nanorings of molybdenum disulfide (MoS₂) nanopores inspired by nuclear pore complexes. We characterize the nanoring of MoS₂ nanopores by scanning transmission electron microscopy, Raman, and X-ray photoelectron spectroscopy spectra. After fabricating MoS₂ nanopore rings-based bio-FET, we confirm edge-selective functionalization by the gold nanoparticle tethering test and the change of electrical signal of the bio-FET. Ultrahigh sensitivity of the MoS₂ nanopore edge rings-based bio-FET (limit of detection of 1 ag/mL) and high selectivity are accomplished by effective coupling of the aptamers on the nanorings of the MoS₂ nanopore edge for cortisol detection. We believe that MoS₂ nanopore edge rings-based bio-FET would provide platforms for everyday biosensors with ultrahigh sensitivity and selectivity.

KEYWORDS: MoS₂ bio-FET, nanopatterning, edge engineering, edge functionalization, ultrasensitive biosensor



INTRODUCTION

The field-effect transistors (FETs) are extensively used as key elements of integrated circuits, computers, communications, and other applications. Especially, with high advantages of label-free, rapid, and ultrasensitive detection of target molecules,^{1–3} FETs are highly desired as sensors, leading to a wide range of applications in medical care,⁴ agriculture,⁵ and environment.⁶ With suitable modifications, biological FETs (bio-FETs) display extraordinarily sensitive interaction between target and receptor elements. In particular, the detection performance of bio-FETs is much affected by the structural and electrical characteristics of the channel materials, thereby indicating that assay performance can be further enhanced for example by introducing nanostructures in the channel.

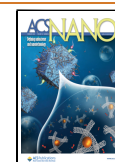
With the advantages of retaining bandgaps compared to the most widely known two-dimensional (2D) nanomaterial, graphene with zero bandgap, transition metal dichalcogenides (TMDs) are considered promising materials for ultrasensitive biosensors.^{7–9} In spite of the intrinsic carrier transport properties, however, the use of TMD materials for applications

in biosensors has been limited due to the restraint in proper functional groups on their surface. The van der Waals force of the TMDs allows a physical adsorption of the receptor biomolecules on their surface.^{10,11} However, these methods induce unstable and random binding of the receptors, resulting in loss of the target binding capacity and denaturation of the biomolecules. Therefore, mostly in TMD bio-FETs, grafting layers such as hafnium oxide (HfO₂),^{7,12,13} aluminum oxide (Al₂O₃),^{14,15} titanium oxide (TiO₂),¹⁶ and electron-beam deposited gold nanoparticles layer¹⁷ have been used to link receptor molecules on top of the TMD channels. However, the use of the interlayer increases the physical distance between the channel and the charged biomolecules and screens the

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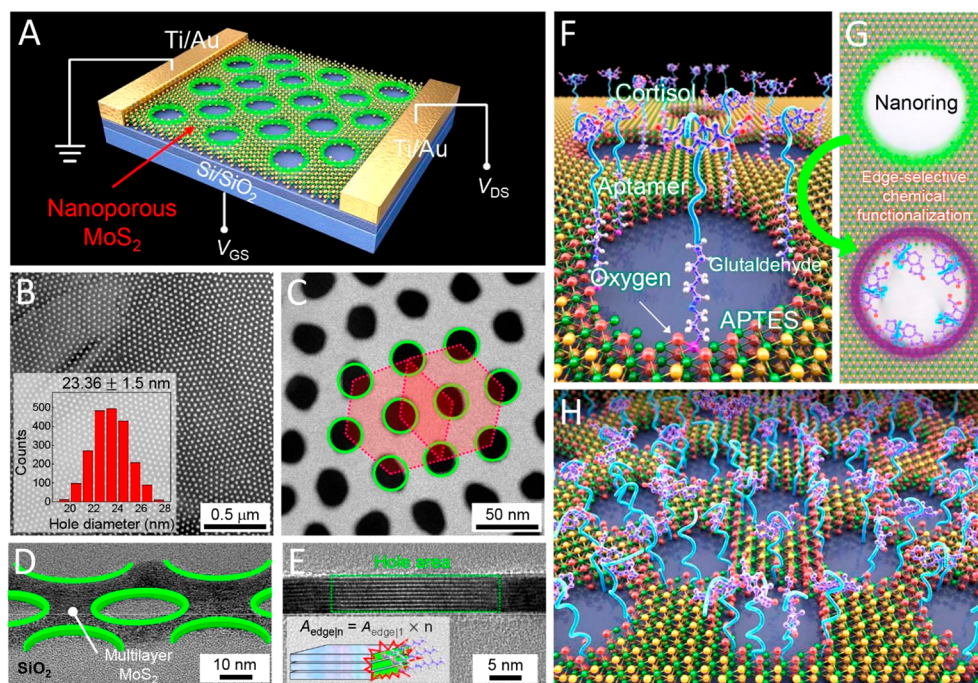


Figure 1. Ultrasensitive and selective bio-FET created by the nanoring edges of MoS₂ nanopores. (A) Schematic diagram of Bio-FET with the nanoring edges of MoS₂ nanopores. (B and C) STEM images of nanoporous MoS₂ films (inset of panel B, statistical distribution of nanopore diameters on the MoS₂). (D and E) Cross-sectional STEM images of MoS₂ nanopore edge with multiple layers (inset of panel E, an increment in edge area by using multilayer MoS₂). (F) Selective edge functionalization of aptamer-cortisol complexes on MoS₂ nanopore edges using APTES–glutaraldehyde chemistry. (G) Enhanced adsorption of aptamer-cortisol complexes on the nanoring of MoS₂ nanopore by selective functionalization. (H) Schematic diagram of multiple nanorings of MoS₂ nanopore edges with aptamer-cortisol complexes.

signaling charges caused by interaction between the receptor molecules and the target analytes,^{18,19} which is manifested as reduced sensitivity. In addition, the interface defects between the channel and grafting layer produce incidental electric fields or parasitic coupling, further contributing to the lowering of sensitivity and reliability in bio-FETs.²⁰

To circumvent the limitation in surface functionality of TMDs, several groups have proposed the bio-FETs using dispersed TMDs, in which the TMD nanoflakes or nanowires were overlapped to form the channel, to introduce abundant edges for the chemical surface modification.^{21,22} However, those devices are not suitable for biosensing electronics due to low stability, high noise level, and biotoxicity derived from the immoderate defects and impurities. Fathi-Hafshejani et al.²³ reported the surface chemical adsorption of the biomolecules using selenium vacancies in tungsten diselenide (WSe₂). However, due to the limited yield in the selenium vacancies, this method is not expected to achieve ultrasensitive biosensors. We presented various TMD bio-FETs with their biomolecule adsorption method and detection property in Table S1.

In this work, we significantly augmented the edge area by adopting nanoporous structure in molybdenum disulfide (MoS₂) inspired by the selective biomolecular interactions of the nuclear pore complex. We applied block copolymer (BCP) lithography to utilize the newly generated dangling groups on the nanoring edges of the nanopores for the selective functionalization of nanorings' surface areas. Periodically arranged nanopores on MoS₂ and corresponding chemical changes in MoS₂ were confirmed by scanning transmission electron microscopy (STEM), Raman, and X-ray photoelectron spectroscopy (XPS) analysis. In particular, the direct

covalent linkage between MoS₂ and the biorecognition elements was expected to manifest a highly sensitive interaction of the bio-FET with target molecules. Indeed, superb detection performance has resulted in a limit of detection (LOD) reacting to 1 ag/mL toward cortisol, both in human serum and artificial saliva.

RESULT AND DISCUSSION

MoS₂ Nanoporous Bio-FET for Sensitive and Selective Detection of Cortisol. Various nanomaterials have been used for high-performance biosensors by designing sophisticated schemes, often entailing complicated procedures.^{24,25} However, in a simple scheme of bio-FET, we accomplished an ultrahigh sensitive detection by introducing nanoring edges of MoS₂ nanopores (Figure 1A). Multilayered MoS₂, an extension of 2D materials, although endowed with intrinsic morphological advantages as a sensor, is unsuitable for applications as biosensors due to the absence of functional groups upon which capture molecules can be attached. In this study, the functionality problem was overcome by generating an innumerable number of nanosized holes of uniform diameter on MoS₂. We established a procedure for fabricating the nanoporous MoS₂ using the self-assembled nanostructure of BCP film as a template (Figure S1). The BCP template's scanning electron microscopy (SEM) image is shown in Figure S2A, indicating the characteristic ordering of nanoporous structure over a large area with a uniform hole diameter (20.7 ± 1.3 nm). The template was treated with oxygen (O₂) plasma reactive-ion etching (RIE) to control the hole size of the BCP template. Before MoS₂ was perforated by boron trichloride (BCl₃) plasma RIE, a protective layer located on the MoS₂ was etched away by sulfur hexafluoride (SF₆) plasma RIE. The

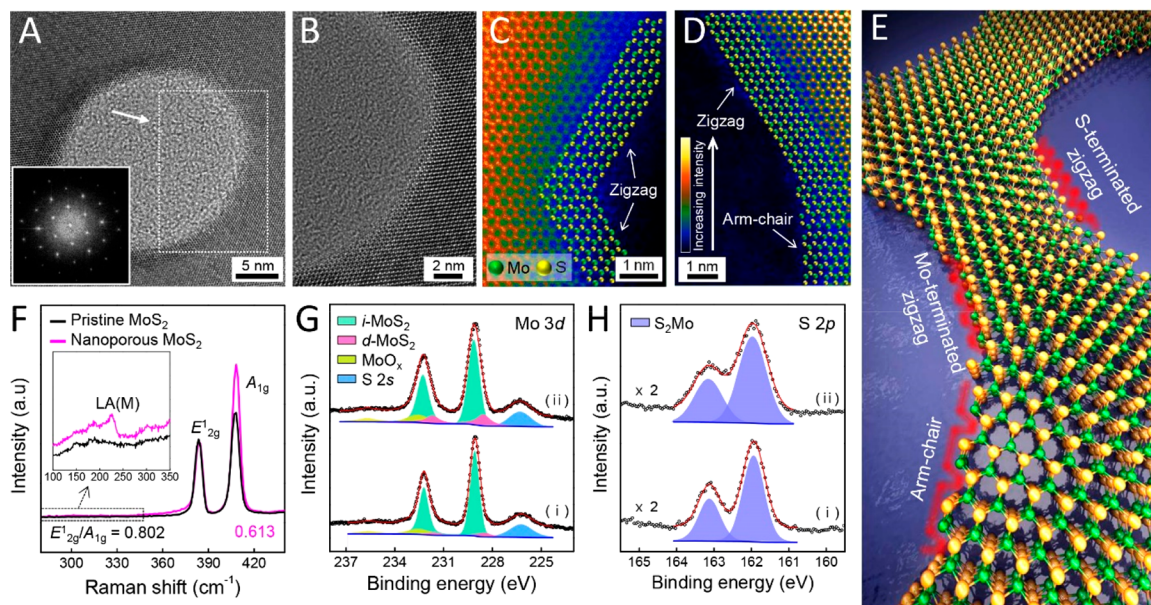


Figure 2. Structural and spectroscopic characterization of the nanoring edges of MoS₂ nanopores. (A) Plan-view STEM image of nanohole with (inset) FFT pattern corresponding to (100)-oriented hexagonal 2H MoS₂ plane. (B) High-magnification STEM image of the marked area in panel A. (C) High-magnification STEM image of zigzag edge of MoS₂ nanopores with intensity mapping profile. (D) Zigzag and arm-chair edges of MoS₂ nanopores with intensity mapping profile. (E) Schematic illustration of edge configurations of MoS₂ nanopores. (F) Raman spectra of pristine MoS₂ and nanoporous MoS₂. XPS spectra of (G) Mo 3d and (H) S 2p core level in multilayer MoS₂ (i) before and (ii) after nanopatterning.

detailed process is described in the Experimental Section and the SEM images of the MoS₂ surface for each process are shown in Figure S2B–D.

Figure 1B shows a low-magnification STEM image of the nanoporous MoS₂, in which nanopores were observed on the whole area of the multilayer MoS₂. The periodically organized nanopores have hexagonally packed ordering, as shown in Figure 1C. The structural uniformity of the nanopores was confirmed by measuring the diameter of 2200 nanopores from three different substrate samples, which was calculated to be 23.36 nm with a standard deviation of 1.5 nm (Inset of Figure 1B). Further, the vertically oriented nanopore structure across the multilayered MoS₂ was confirmed by a cross-sectional STEM image, as shown in Figure 1D,E. In Figure 1E, the nanopore edge areas look brighter than the unperforated areas due to differences in the distance from the surface MoS₂ atoms to the objective lens of the STEM. In the brighter nanopore area, 10 MoS₂ layers can be clearly observed. The edge area (A_{edge}) can be significantly increased by an increment of the layer numbers (n), facilitating an important enhancement in edge functionalization efficiency.

Figure 1F shows a conceptual illustration of the nanoring edge functionalization of cortisol aptamer and the detection of cortisol on the MoS₂ nanopores. The nanoring edges are oxidized by O₂ plasma treatment and functionalized with cortisol aptamer using the well-known (3-aminopropyl) triethoxysilane (APTES)–glutaraldehyde chemistry.¹⁵ Thermodynamically unstable edges are easily functionalized with the chemical groups and a large number of aptamer–cortisol complexes are selectively captured on the nanoring edges of the MoS₂ nanopore (Figure 1G). The nanoring edges provide abundant binding sites for the cortisol aptamers, further leading to a significant increase in capturing of the cortisol molecules (Figure 1H). Those abundant biomolecular binding sites on the MoS₂ obtained by a myriad of nanoporous

structures are thought to contribute to the concomitant augmentation of sensing performance in the nanoporous bio-FETs. In addition, in contrast to the previous MoS₂ bio-FETs where oxide layers were used for grafting probe molecules since no additional conjugating layer is used for the nanoporous MoS₂, the carrier modulation effect of the negative charge on cortisol seems to proceed in a highly efficient manner in the absence of the screening of the signal charge.

Spectroscopic Analysis of Nanoporous MoS₂. The atomic structure of MoS₂ nanoring was observed by STEM measurement. Figure 2A shows a highly magnified STEM image of the nanoring of the MoS₂ nanopore with a close-up view of the dashed area presented in Figure 2B. The fast Fourier transform (FFT) pattern extracted from Figure 2A suggests (100)-oriented hexagonal 2H MoS₂ film with AA' stacking (inset of Figure 2A). To investigate the atomic configuration of the circular nanoring edges, an atomic-scale STEM observation of two sides of the circular nanoring was performed (Figures 2C,D). Indeed, the intensity mapping images exhibit anisotropically etched edge structures of the perforated MoS₂ multilayer. The atomic configurations of the top MoS₂ layer show both zigzag and armchair arrangements of Mo and S atoms at the nanoring edges. The atomic structures of each edge configuration are described in a schematic image of the nanopore in Figure 1E. Zigzag edges are classified into Mo-terminated and S-terminated zigzag according to the type of outermost atom. However, Mo-termination and S-termination are repeated layer by layer due to the AA' stacking sequence of the MoS₂ films.

To study the effects of the newly formed morphological changes in MoS₂, a comparative analysis was performed for both the nanoporous and pristine MoS₂ films using Raman and XPS. In the Raman spectra (Figure 2F), two characteristic Raman peaks of the in-plane, E'_{2g} , and out-of-plane, A_{1g} , vibrations were observed at 380 and 407 cm^{−1}, respectively, in

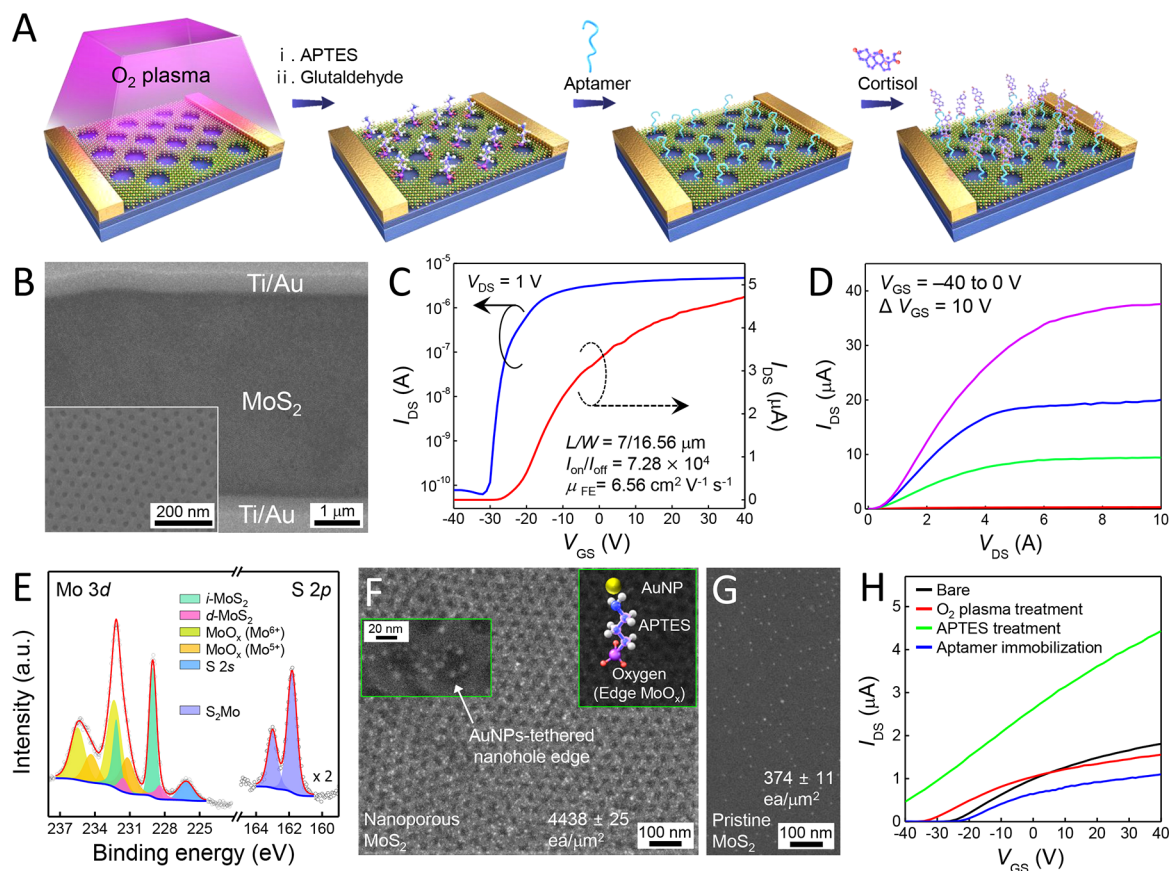


Figure 3. Preparation of MoS₂ bio-FET with nanopore edge for selective biomolecular detection. (A) Experimental process of functionalization of aptamer on the surface of nanoporous MoS₂ and detection of cortisol. (B) SEM images of nanoporous MoS₂ FET surface. (C) Transfer and (D) output characteristics of nanoporous MoS₂ FET. (E) XPS spectra of nanoporous MoS₂ after O₂ plasma RIE treatment. SEM images of (F) nanoporous MoS₂ and (G) pristine MoS₂ surface tethered with AuNPs of 5 nm on APTES-treated surface. (H) Transfer curves of the nanoporous MoS₂ bio-FET measured during functionalization processes.

both the pristine and the nanoporous MoS₂.²⁶ After perforation, the relative intensities of the in-plane and out-of-plane vibrations (E_{2g}^1/A_{1g}) decreased from 0.802 for the pristine to 0.613 for the nanoporous MoS₂. The lowering of E_{2g}^1/A_{1g} in the nanoporous MoS₂ well supports the presence of the newly generated nanopores and increase in the edge sites since A_{1g} vibration is preferentially excited to the E_{2g}^1 vibration for edge-terminated films.^{27,28} In addition, a new intense peak can be seen at $\sim 227\text{ cm}^{-1}$ for the perforated MoS₂ (inset of Figure 2F). This band is a longitudinal acoustic (LA) branch at the M point of the Brillouin zone. It corresponds to disorder-induced Raman scattering in MoS₂, similar to the D band in defective graphene.^{29,30} This band was activated on the nanoporous MoS₂ film because the edge defects were introduced into the MoS₂ lattice.

Figure 2G shows XPS analysis of the pristine (i) and the nanoporous (ii) MoS₂ films concerning their Mo 3d core levels. The corresponding S 2p spectra are shown in Figure 2H. In the Mo 3d spectra, the peak was deconvoluted into three different types of Mo ligands corresponding to the intrinsic MoS₂ (*i*-MoS₂), defective MoS₂ (*d*-MoS₂), and molybdenum oxide (MoO_x).^{31,32} In the doublet of *i*-MoS₂, the maximum peak at around 229.10 eV (Mo⁴⁺ 3d_{5/2}), which corresponds to 2H stoichiometric MoS₂ (ratio of S/Mo = 2), was observed in both the pristine and nanoporous MoS₂ films with high intensity. The maximum peak of the *d*-MoS₂ at 228.50 eV (Mo⁴⁺ 3d_{5/2}) is relatively small, corresponding to the

nonstoichiometric MoS₂ (S/Mo ratio <2) introduced by atomic defects on the film, such as vacancies and exposed edges.³¹ These defect sites have electronic structures different from those of the intrinsic MoS₂ owing to the unstable energy state, resulting in peak excitation at lower binding energies. The atomic fraction of *d*-MoS₂ among the total Mo ligands was calculated to be 6.18% for the pristine MoS₂ film, which significantly increased to 15.82% for the nanoporous MoS₂ film (Table S2). This can be interpreted as the quantitative increase in the edge sites from the nanoporous MoS₂. In addition, the increase in nonstoichiometric MoS₂ in the nanoporous MoS₂ caused the characteristic broadening of the S 2p doublet, with a concomitant increase in the full width at half-maximum (fwhm) from 0.68 to 0.89. Contaminations on the active edge sites may occur during the etching process with BCl₃ plasma RIE. However, the absence of a Cl 2p peak in the nanoporous MoS₂ film (Figure S3) indicates that no contamination occurred during the BCl₃ etching process. In addition, the atomic fraction of Mo⁶⁺ 3d (MoO_x) at approximately 232.53 eV barely changed from 11.93% for the pristine to 11.49% for the nanoporous MoS₂, indicating that the active edge regions were not oxidized on exposure to the atmosphere.

Preparation of Nanoporous MoS₂ Bio-FET and Surface Functionalization for Cortisol Detection. Cortisol is a glucocorticoid steroid hormone secreted through the hypothalamus–pituitary–adrenal (HPA) axis. Repeated activation of the HPA axis has been reported to negatively affect

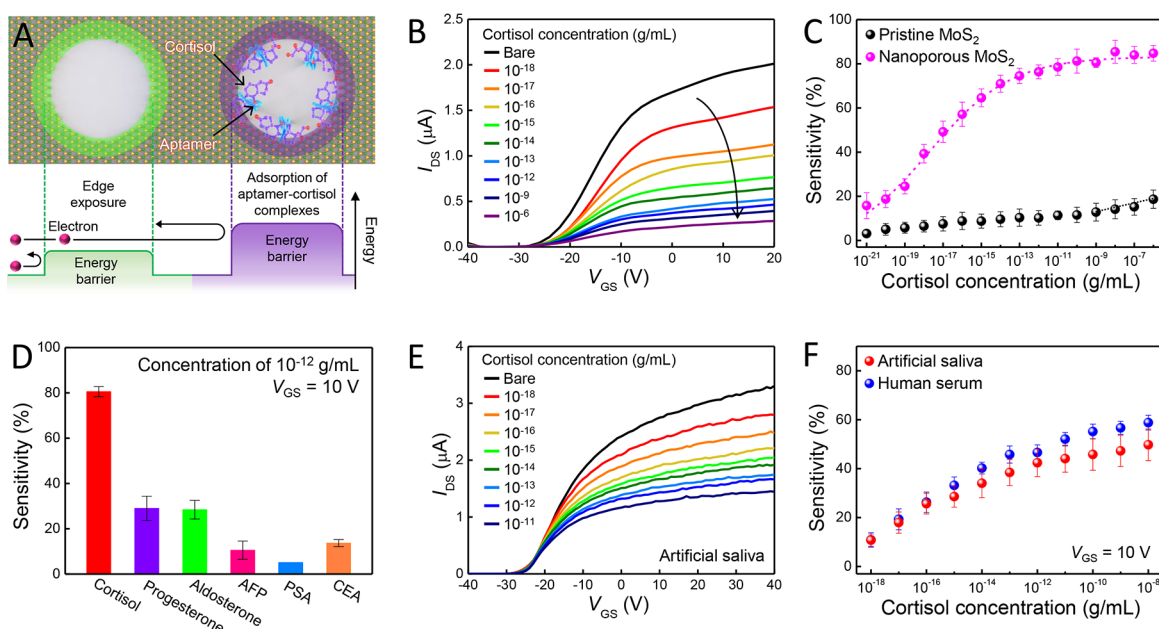


Figure 4. Application of MoS₂ nanopore edge-based bio-FETs for ultrasensitive and selective detection of cortisol. (A) Schematic images of MoS₂ nanopore and aptamer-cortisol adsorbed-nanopore, respectively, and related energy barrier height. (B) Transfer characteristics of the aptamer-immobilized on the nanoring edges of MoS₂ nanopore-based bio-FET with increasing cortisol concentrations. (C) A comparison of sensitivity of the pristine and the nanoporous MoS₂ bio-FETs according to cortisol concentrations. (D) Selectivity of the nanoporous MoS₂ bio-FETs for cortisol detection with several potentially interfering steroid hormones and antigens. (E) Detection characteristics of the nanoporous MoS₂ bio-FET for different concentrations of cortisol in artificial saliva. (F) Sensitivity of the bio-FETs in artificial saliva and real human serum.

mental health, causing major depressive disorder,³³ anxiety disorder,³⁴ and bipolar disorder.³⁵ The performance of the nanopatterned bio-FET was confirmed by studying its detection behavior toward cortisol. Figure 3A describes a general procedure for constructing the nanoporous MoS₂ bio-FET, in which a MoS₂ FET was treated with O₂ plasma and functionalized with cortisol aptamer using APTES-glutaraldehyde chemistry. Figure 3B is a SEM image of the MoS₂ FET presenting uniform nanopores across the channel area positioned between the two Ti/Au electrodes. Figure 3C depicts the transfer characteristics of the nanoporous MoS₂ FET in logarithmic (blue line), and linear (red line) scales measured under a gate voltage (V_{GS}) from -40 to 40 V at a drain voltage (V_{DS}) of 1 V. The length (L) and width (W) of the channel were 7 and 16.56 μm , respectively. It demonstrated an n -type semiconductor behavior with an on-off ratio (I_{on}/I_{off}) of 7.28×10^4 and field-effect mobility (μ_{FE}) of 6.56 $\text{cm}^2 \text{V}^{-1} \text{s}^{-1}$. The output characteristics of the n -type nanoporous MoS₂ FET were shown in Figure 3D. We manufactured 20 different nanoporous MoS₂ FETs prepared from 20 different nanopatterned MoS₂. The statistical distribution of μ_{FE} for those MoS₂ FETs is depicted in Table S3, together with the values for I_{on}/I_{off} . The relatively low values of μ_{FE} in the nanoporous MoS₂ FET, compared to those ($\sim 100 \text{ cm}^2 \text{V}^{-1} \text{s}^{-1}$) of the typical pristine multilayer MoS₂ FETs reported in the literature,³⁶ are thought to have been caused by an increase in carrier trapping and scattering in the MoS₂ channel due to increased defective edge areas of the nanopores.³⁷ However, stability of the bio-FETs turned out to be sound, as was confirmed by retaining the electrical performances even after repeated exposure to ambient air and phosphate-buffered saline (PBS) liquid (Figure S4A,B).

Treatment of O₂ plasma on the nanoporous MoS₂ FET was processed under mild conditions (5 sccm, 10 W, 15 s) since MoS₂ is prone to be damaged when exposed to O₂ plasma, which in turn may manifest into a detrimental effect on the electrical properties of the bio-FET. To understand the effect of the oxidation treatment, XPS analysis was carried out for both the nanoporous (Figure 3E) and the pristine (Figure S5) MoS₂. When O₂ plasma was treated for the nanoporous MoS₂, the atomic fraction of Mo⁴⁺ 3d was significantly reduced from 72.70% to 32.30% for i -MoS₂ and from 15.82% to 4.90% for d -MoS₂, respectively (Table S4). The atomic ratio of S²⁻ 2p to Mo⁴⁺ 3d also decreased from 2.04 to 1.87, signifying that the O₂ plasma treatment broke the bonding between Mo and S and replaces O at the location where S was (Table S4). With the generation of a new intense peak of MoO_x (Mo⁵⁺ 3d) at 231.33 eV, the total amount of MoO_x (Mo⁶⁺ 3d and Mo⁵⁺ 3d) increased from 11.49% to 62.80%. However, when the same treatment was done for pristine MoS₂, only a negligible amount of variation in the atomic ratio of S²⁻ 2p to Mo⁴⁺ 3d was observed from 2.06 to 2.01. From these, it can be conjectured that the oxidation in nanoporous MoS₂ occurs selectively at the newly generated edge regions in the nanoporous structure. However, the increase in MoO_x (Mo⁶⁺ 3d and Mo⁵⁺ 3d) peaks in the pristine corresponds to the relatively unstable oxide layer (MoS_{2-y}O_y)³⁸ on the MoS₂ basal plan. To find out if immobilization of the cortisol aptamer could be processed in the oxidized edge area of nanoporous MoS₂, possibly owing to the APTES chemistry, we investigated an assembly of 5 nm gold nanoparticles (AuNPs) on the APTES-functionalized nanoporous MoS₂ film (Figure 3F). For comparison, the same experiment was conducted on the pristine MoS₂ surface (Figure 3G). The density of AuNPs was measured to be $374 \pm 11 \text{ ea}/\mu\text{m}^2$ for the pristine MoS₂ surface,

which was enhanced significantly to $4438 \pm 25 \text{ ea}/\mu\text{m}^2$ for the nanoporous MoS_2 surface. In addition, it is noteworthy that the AuNPs were tethered favorably on the edge sites of nanopores rather than on the basal plane exposed between the nanopores (inset of Figure 3F), which indicates the efficient edge-selective functionalization using the APTES chemistry on the nanoporous MoS_2 . Many papers have hypothesized the selective adsorption of analyte molecules at the edge based on the rich dangling bonds of the edge area.^{39–41} However, we intuitively demonstrated the edge-selective functionalization through the AuNPs tethering experiment.

Figure 3H shows transfer curves of the nanoporous MoS_2 bio-FET measured during the functionalization process. The formation of MoO_x on edge induced the decrease in drain current (I_{DS}) at the on-state ($V_{\text{GS}} > 0 \text{ V}$) due to p -type doping effect,^{42,43} whereas the threshold voltage (V_{TH}) was negatively shifted due to an increase of the depletion mode as $\text{MoS}_{2-y}\text{O}_y$ was formed on the basal plane.³⁸ A significant increase in I_{DS} was also observed after the APTES treatment due to the n -type doping effect of APTES,⁴⁴ and this effect was negated when the cortisol aptamer was immobilized via glutaraldehyde treatment.

Sensing Performance of Nanoporous MoS_2 Bio-FET for Cortisol. Cortisol takes a negative charge under the measurement condition (pH 7.4) since the isoelectric point of cortisol is around 5.2.^{45,46} The nanopore on the MoS_2 lattice provides energy barriers due to the conductivity difference between the MoS_2 basal plane and edge area (green hole and the corresponding energy diagram in Figure 4A).^{37,47} The low conductivity of the edge regions is derived from scattering and trapping of mobile carriers at the newly generated trap states. We studied the generation of trap states in the nanoporous MoS_2 structure in a previous work.⁴⁸ When negatively charged cortisols are adsorbed at the nanoring edge area, the negative potential is accumulated on the MoS_2 edge area and induces depletion of the electron carriers by inhibiting the electron migration in the channel, further raising the energy band at the edge area.^{49–51} Eventually, it seems to result in a higher energy barrier (purple hole and the corresponding energy diagram in Figure 4A). The sensing behavior of the nanoporous MoS_2 bio-FET was confirmed by gradually adding the precalculated amount of cortisol in a stepwise manner, so that the target concentration could be adjusted as required. Highly sensitive variations of transfer curves were observed in response to varying amounts of cortisol, as shown in Figure 4B. An increase in cortisol concentrations from 10^{-18} g/mL to 10^{-6} g/mL results in a gradual decrease of I_{DS} values due to increased energy barrier height.

To understand the effect of nanoring edges of MoS_2 nanopores on the sensing behavior, the sensing characteristics of the nanoporous and pristine MoS_2 FETs were compared (Figure 4C). The pristine MoS_2 bio-FETs were prepared by following the same surface treatment used for the nanoporous bio-FETs as described in Figure 3A. The sensing behavior of bio-FET with pristine MoS_2 is shown in Figure S6A, and current variations ($|I_{\text{bare}} - I_{\text{cortisol}}|$) of the pristine MoS_2 were compared with those of nanoporous MoS_2 bio-FET in Figure S6B, where I_{bare} and I_{cortisol} are the I_{DS} values before and after the addition of cortisol, respectively, at a V_{GS} of 10 V. The sensitivity was calculated based on $(|I_{\text{bare}} - I_{\text{cortisol}}|)/I_{\text{bare}} \times 100$, at a V_{GS} of 10 V. A rather typical S-shaped response pattern was observed for the nanoporous MoS_2 FETs with a steep linear mode observed between 10^{-18} and 10^{-14} g/mL in

cortisol concentration, while a slight hint of plateauing behavior was revealed at a low concentration range. In contrast, for the pristine MoS_2 FETs, no such apparent shape pattern was observed with only a slight hint of slope mode starting at around 10^{-9} g/mL of cortisol concentration. The two curves of the sensing behavior well support the surprisingly huge amount of enhancement effect in sensing performance of the MoS_2 nanoporous structure at least up to 10^9 times that of the pristine MoS_2 .

The reliability of the nanoporous bio-FET was examined by studying the extent of nonspecific binding and selectivity using potentially interfering biomolecules. To understand the level of nonspecific interaction that might have been generated due to nanopore formation in MoS_2 , cortisol detection was performed using nanoporous FETs in the absence of the aptamer capture molecules, specifically the nanoporous MoS_2 FET in which the surface chemistry was altered only up to the functionalization step of glutaraldehyde. No appreciable signal change was detected (Figure S7), suggesting that nonspecific interaction was negligible.

In addition, the selectivity of the nanoporous MoS_2 bio-FET for target cortisol was evaluated by measuring the electrical signals on exposure to potentially interfering biomolecules, including steroid hormones of progesterone and aldosterone, and protein biomarkers of alpha-fetoprotein (AFP), prostate specific antigen (PSA), and carcinoembryonic antigen (CEA) (Figure 4D). The sensitivity values after exposure to the protein antigens and the steroid hormones were noticeably lower than that of the cortisol. However, in the case of the steroid hormones, relatively increased sensitivity values were observed than for the proteins. It is worthwhile to note that the chemical structure of cortisol is very similar to that of progesterone and aldosterone.

One major advantage of using cortisol as a biomarker is the diverse availability of its biological fluids, including serum, urine, sweat, and saliva. In particular, measuring cortisol concentration in saliva has considerable merits, including being a patient-friendly noninvasive detection method and direct measurement of free unbound cortisol, which is a biologically active form; 90% of cortisol circulates in complexation with globulin.⁵² To study the applicability of the nanoporous MoS_2 bio-FET in the clinical environment, the detection behavior of the bio-FET was examined in the media of artificial saliva and human serum. Figure 4E shows $I_{\text{DS}}-V_{\text{GS}}$ curves when the detection test was performed by varying the amount of cortisol from 10^{-18} g/mL to 10^{-8} g/mL in artificial saliva including the control test. Well-separated signals are evident, especially at a low concentration range with a gradual decrease in distance between signals at a higher concentration window. This tendency was somewhat more pronounced in artificial saliva than in human serum (Figure 4F). The $I_{\text{DS}}-V_{\text{GS}}$ curves for real human are shown in Figure S8. Compared with the results from the buffer test, the detection performance in the artificial saliva and serum seemed to have decreased to some extent, especially in the low concentration range, due to an inhibiting effect of innumerable types of biological entities and salt components contained in the saliva and serum. Those coexisting species adsorb onto a sensing interface nonspecifically and cause an adverse effect in the interaction of the probe molecules with the target molecules.^{53,54} However, the LOD remained at 10^{-18} g/mL with linearity ranging up to 10^{-12} g/mL in those biological fluids. We solidified the excellent performance of the nanoring MoS_2 bio-FETs by comparing

with various types of cortisol sensors in terms of sensitivity and selectivity, as shown in Table S5. The nanoring MoS₂ bio-FETs offer significant advantages over other sensors with a wide detection range, ultralow LOD, and superb selectivity.

CONCLUSION

We accomplished an ultrasensitive and selective bio-FET by the nanorings of MoS₂ nanopores inspired by nuclear pore complexes. Via BCP nanotemplates, periodically organized nanopores were constructed on the MoS₂, thus providing abundant nanoring edges around the nanopores. While the generation of nanoring edge sites on the MoS₂ surface resulted in degradation of the electrical performance for FET, ultrahigh sensitivity was accomplished as a bio-FET due to enabling functionalization on the newly generated edges in the nanopatterned MoS₂. In addition, the nanoporous bio-FET functionalized with aptamer exhibited excellent selectivity for cortisol compared to other steroids and antigens, including progesterone, aldosterone, AFP, PSA, and CES. Clinical applicability was also confirmed by performing the test on artificial saliva and actual human serum. We confirmed the enhanced performance of the nanoporous MoS₂ bio-FET by realizing an ultrasensitive LOD of 1 ag/mL. With such a superb detection performance, the nanopatterned MoS₂ bio-FET may be applied for polymerase chain reaction-free detection of biomarkers.

EXPERIMENTAL METHODS

Fabrication of Nanoporous MoS₂ FET. The general fabrication process of the nanoporous MoS₂ consists of organizing the BCP nanopattern on MoS₂, etching the MoS₂, and subsequently removing the upper BCP layers. Multilayer MoS₂ nanosheets were placed on a Si/SiO₂ substrate by mechanical exfoliation from bulk MoS₂. A 10 nm thick SiO₂ layer was deposited onto the MoS₂ nanosheets using an electron-beam (e-beam) evaporator. A 1 wt % solution of poly(styrene-*r*-methyl methacrylate) (PS-*r*-PMMA) ($M_n = 8500$ $M_w/M_n = 1.45$, styrene content of 66%, α -hydroxyl- ω -TEMPO moiety terminated) random copolymer (RCP) in toluene was spin-coated on the MoS₂ films at 3000 rpm, followed by substrate annealing at 250 °C for 2 h under vacuum and finally rinsing with toluene. RCP was used as a grafting layer to produce neutral and uniform wetting of the BCP film on the substrate. A solution of 1 wt % poly(styrene-*b*-methyl methacrylate) (PS-*b*-PMMA) (PS/PMMA = 55 000:22 000, $M_w/M_n = 1.09$) BCP in toluene was also spin-coated with the same protocol as RCP and annealed at 230 °C. The PMMA portions in the BCP thin film were decomposed by UV irradiation (wavelength of 254 nm) for 30 min and finally dissolved in acetic acid solution. O₂ plasma RIE (10 sccm, 10 W, 10 s) was applied to the surface of BCP. SF₆ plasma RIE (10 sccm, 200 W, 15 s) was carried out to punch nanopores on the SiO₂ layer. Subsequently, BCl₃ plasma RIE (10 sccm, 100 W) was used to make MoS₂ nanopores, with the plasma treatment time-varying with the thickness of MoS₂. At the end of the process, the substrate was submerged in BOE for 1 s to eliminate the remaining SiO₂ and BCP on the MoS₂. To construct the FET on the prepared nanoporous MoS₂, the source and drain were prepatterned by photolithography and a lift-off process, followed by deposition of Ti (20 nm)/Au (100 nm) by e-beam evaporation. Pristine MoS₂ bio-FET was fabricated by the deposition of the Ti/Au electrode on exfoliated MoS₂.

Characterizations. Microstructural studies of the nanoporous MoS₂ were conducted using STEM (HD-2300A, Hitachi) with an accelerating voltage of 200 kV. For plane-view observation, the nanoporous MoS₂ films were transferred onto Cu grids coated with a lacey carbon film. A cross-sectional view was observed by milling the sample using a single-beam focused ion-beam (FB-2100, Hitachi). Low-magnification surface images were obtained by SEM operated at

an acceleration voltage of 15 kV and working distance of 8 mm. Raman spectroscopy (ALPHA300, WITec) was used to identify the formation of edge sites on MoS₂ nanosheets. A laser beam with a spot diameter of 1 μ m and excitation wavelength of 532 nm was used, and the instrument spectral resolution was approximately 1 cm⁻¹. The chemical states of the MoS₂ films were explored by XPS (K-Alpha, Thermo Fisher Scientific) using an Al K α source. All electrical measurements were carried out using a semiconductor characterization analyzer (4200-SCS, Keithley) equipped with a probe station for sample loading and electrode contact.

MoS₂ Surface Modification and Biosensing Process. Specific detection of cortisol was facilitated by functionalizing an aptamer, which is a bioreceptor for cortisol, on the channel surface of the nanoporous MoS₂ bio-FET. The aptamer (Bioneer) was dissolved in Tris-HCl (pH 8.0) solution to obtain a concentration of 10⁻⁶ g/mL. The MoS₂ bio-FET was first exposed to O₂ plasma (10 sccm, 10 W, 15 s) for edge oxidation. The MoS₂ was then functionalized with APTES by treating with a solution of 0.4 mL of APTES in a 19:1 (v/v) mixture of ethanol and deionized (DI) water for 3 h. Subsequently, the device was rinsed in ethanol and dried at 120 °C for 15 min. To convert the amine functional group of APTES to an aldehyde group, the device was immersed in 4.5 mL of glutaraldehyde solution consisting of 0.1 g of NaCNBH₄, 1 tablet of PBS, and 200 mL of DI water, for 2 h, followed by rinsing in DI water. The device was incubated overnight in a 10⁻⁶ g/mL aptamer solution at 4 °C. To prevent nonspecific binding, 1% (w/v) casein blocker (Thermo Fisher Scientific) was added to the device for 1 h at room temperature. For the detection of cortisol, 16 different concentrations of cortisol solution (Sigma-Aldrich) ranging from 10⁻²¹ g/mL to 10⁻⁶ g/mL were dissolved in PBS solution (pH 7.4). Each cortisol solution was dropped onto the device for 30 min and subsequently rinsed and dried for electrical measurements.

Detection in Saliva and Human Serum. To understand the sensing behavior in the biological environment of saliva and serum, different concentrations of cortisol were dissolved in artificial saliva and real human serum. Real human serum was purchased from Sigma-Aldrich. Artificial saliva was prepared by dissolving cellulose, 5 mM NaCl, 1 mM CaCl₂, 15 mM KCl, 1 mM citric acid, 1.1 mM KSCN, and 4 mM NH₄Cl in distilled water.

ASSOCIATED CONTENT

Supporting Information

The Supporting Information is available free of charge at <https://pubs.acs.org/doi/10.1021/acsnano.1c08255>.

Comparison of the nanoporous MoS₂ bio-FET with other 2D TMDs bio-FETs; fabrication method and SEM images of MoS₂ surfaces for each fabrications processes; additional XPS analysis data of the nanoporous MoS₂ films; statistical distribution of electrical characteristics of the nanoporous MoS₂ bio-FETs and device stability tests in air and liquid environments; comparison of XPS spectra between the pristine and nanoporous MoS₂ films before and after O₂ plasma treatment; additional supporting information on the biomolecular sensing performances of the nanoporous MoS₂ bio-FETs; comparison of sensing performances of several cortisol biosensors with the nanoporous MoS₂ biosensor (PDF)

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Author Contributions

[†]H.P. and S.B. contributed equally. L.P.L., Y.J.K., and S.K. conceived the project. H.P., and S.B. fabricated the nano-patterned MoS₂ bio-FETs and measured the electrical properties and sensing properties of the devices. A.S., and J.S. supported the experiments. B.J. provided the bimolecular resources. Y.C.P. contributed to structural characterizations of the MoS₂ films using STEM analysis.

Notes

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